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# Degradation, Bone Regeneration and Tissue Response of an Innovative Volume Stable Magnesium-Supported GBR/GTR Barrier Membrane

## Dissertation

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# Degradation, Bone Regeneration and Tissue Response of an Innovative Volume Stable Magnesium-Supported GBR/GTR Barrier Membrane

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Abstract: Introduction: Bioresorbable collagenous barrier membranes are used to prevent premature soft tissue ingrowth and to allow bone regeneration. For volume stable indications, only non-absorbable synthetic materials are available. This study investigates a new bioresorbable hydrofluoric acid (HF)-treated magnesium (Mg) mesh in a native collagen membrane for volume stable situations. Materials and Methods: HF-treated and untreated Mg were compared in direct and indirect cytocompatibility assays. In vivo, 18 New Zealand White Rabbits received each four 8 mm calvarial defects and were divided into four groups: (a) HF-treated Mg mesh/collagen membrane, (b) untreated Mg mesh/collagen membrane (c) collagen membrane and (d) sham operation. After 6, 12 and 18 weeks, Mg degradation and bone regeneration was measured using radiological and histological methods. Results: In vitro, HF-treated Mg showed higher cytocompatibility. Histopathologically, HF-Mg prevented gas cavities and was degraded by mononuclear cells via phagocytosis up to 12 weeks. Untreated Mg showed partially significant more gas cavities and a fibrous tissue reaction. Bone regeneration was not significantly different between all groups. Discussion and Conclusions: HF-Mg meshes embedded in native collagen membranes represent a volume stable and biocompatible alternative to the non-absorbable synthetic materials. HF-Mg shows less corrosion and is degraded by phagocytosis. However, the application of membranes did not result in higher bone regeneration.

**Keywords:** barrier membrane; GBR/GTR membrane; degradation; magnesium; collagen; in vivo; histomorphometry; tissue reaction

#### 1. Introduction

Bony defects, resulting from trauma, tumor resection, malformations or progressive degeneration, often require difficult reconstructive bone augmentation procedures. The application of guided

bone-/guided tissue-regeneration (GBR/GTR) technique is a common method to realize bone regeneration in the alveolar ridge area [1–3]. The main principle of GBR/GTR is based on a barrier membrane between the defect and outer soft tissue to prevent rapid soft tissue ingrowth and to promote osseous regeneration [4–9]. In this context, a barrier membrane has to fulfil the five postulated criteria of Scantlebury: tissue integration, cell occlusivity, clinical manageability, space-making and biocompatibility [10].

For dental application, resorbable and non-resorbable membranes of different origins can be distinguished [8]. Furthermore, natural and synthetic materials are available [8]. For indications that do not require volume stability, collagen-based membranes have shown to be optimal biomaterials due to their biocompatibility and degradability that are conducted by cells of the collagen metabolism such as fibroblasts, macrophages and eosinophils [11–14]. However, conventional collagen membranes are usually unstable and can either collapse into the defect or deform under load, resulting in the impairment of tissue regeneration [12,13]. Thus, multidimensional bony defects require the application of volume-stable load-bearing membranes [15–17]. For this indication, non-resorbable materials based on polytetrafluorethylene (e-PTFE) or dense polytetrafluorethylene (d-PTFE) combined with titanium meshes are the gold standard [18,19]. However, the use of non-resorbable materials always requires a second surgical procedure for removement, which can comprise higher overall morbidities, patient discomfort and higher costs [12,13,19]. Besides slightly increased inflammation, the use of non-resorbable membranes can cause bacterial colonization in the event of oral exposure with concomitant regeneration impairment [20–23].

Thus, the clinical need to develop a next generation of volume-stable barrier membranes that combine structural stability with bioresorbable properties is of great interest. Based on the non-resorbable PTFE membranes with titanium meshes, the development of a resorbable collagen membrane with an integrated resorbable metallic mesh is obvious.

In this context, magnesium and a variety of its alloys have been a subject of recent investigations as a degradable implant material [24–26]. Beside its favorable bone-like properties, magnesium is fully degradable in aqueous solutions and biocompatible in natural doses [25,27–29]. However, the rapid uncontrolled degradation along with hydrogen gas evolution, environmental alkalization and rapid structural integrity loss can lead to necrosis and emphysema, and therefore limits a clinical application so far [25,30]. A variety of strategies have been developed to control magnesium degradation and to improve biocompatibility (Figure 1) [31]. Thereby, passivation of the surface can be achieved by conversion (e.g., plasma electrolytic oxidation, ion implantation) and deposited coatings (e.g., chemical vapor deposition, plasma spraying) of the surface [24–26,31,32].



Figure 1. Common strategies to passivate magnesium for in vitro/in vivo application.

As part of the conversion coatings, hydrofluoric acid (HF) treatment is a promising surface passivation method [33–37]. Thereby, hydroxides, oxides and compounds of the general formula  $Mg(OH)_xF_{2-x}$  are formed on the magnesium surface during processing [37,38]. Interestingly, the corrosion rates were up to 20 times lower than those of untreated samples [37,38].

Considering these aspects, the present study aims to investigate a newly developed volume-stable barrier membrane composed of a native collagen membrane with an integrated magnesium mesh for GBR/GTR treatment. HF-treatment of magnesium was applied for passivation aspects and to slow down biodegradation. The newly developed membrane was investigated by established in vitro methods and an in vivo calvarian implantation model using New Zealand White Rabbits [39,40]. Tissue regeneration and reaction were analyzed using established histological, histopathological and histomorphometrical methods [19,41–43].

#### 2. Results

#### 2.1. Manufacturing of a Magnesium Collagen Membrane

Both in vitro and in vivo test samples could be successfully manufactured. Figure 2 displays the HF-treated and untreated magnesium sheets and meshes. The HF-coated samples are characterized by a matte and textured surface, while the untreated sheets have a more brilliant surface. Figure 3 shows the integration of the mesh into the collagen membrane as well as its formability and adaptability.



**Figure 2.** Comparison of untreated and processed samples. (**A**) Pictures of both an untreated and HF-treated magnesium sheet for the in vitro experiments. (**B**) Processed magnesium meshes before embedded into a collagen membrane.



**Figure 3.** Different membrane modifications. (**A**) Integration of the magnesium mesh into the collagen fleece. (**B**,**C**) Moldability and stability of the collagen membrane with an integrated Mg mesh.

#### 2.2. Cytocompatibility Analysis

Generally, values >70% of the negative control in the XTT and BrdU assays and values <130% of the negative control in the LDH assay indicate the nontoxic range as defined in ISO 10993-5:2009 [29]. In both

the XTT and BrdU assays, HF-treated magnesium exhibited suitable cytocompatibility, whereas the untreated magnesium was within the toxic range (Figure 4A–C). Thereby, HF-treated magnesium was not significantly different compared to the negative control in the BrdU assay (p > 0.05), but significantly different from the negative control in the XTT assay ( $p \le 0.05$ ). Both the untreated magnesium and positive control groups were significantly different to the negative control ( $p \le 0.01$ ). In the LDH assay, both magnesium groups were inside the toxic range, while only the HF-treated magnesium group was not significantly different compared to the negative control (p > 0.05). Morphological assessment of the magnesium test samples after 72 h of extraction showed an enhanced corrosion of the untreated test samples, while the HF-treated specimens showed no visible signs of corrosion or degradation (Figure 4D)



**Figure 4.** Cytocompatibility results using L9292 cells of the different variants. (**A**) proliferation measured by a BrdU assay; (**B**) viability measured by a Sodium 3,3'-[1(phenylamino)carbonyl]-3,4-tetrazolium]-3is(4-methoxy-6-nitro) Benzene Sulfonic acid Hydrate (XTT)-assay; (**C**) cytotoxicity measured by a Lactate Dehydrogenase (LDH) assay. Values are either normalized against positive controls (LDH) or negative control (XTT, BrdU). Means with error bars indicating standard deviations. The dotted line indicates thresholds which should not be exceeded (LDH) or fall below (XTT; BrdU). Significant differences are indicated (•: p < 0.05, \*\*: p < 0.01). (**D**) Both untreated and HF-treated magnesium after 72 h extraction. The untreated magnesium shows enhanced corrosions, which is visible due to the high surface porosity and black textured corrosion products. The HF-treated magnesium was not different from the initial morphology.

In the live-dead staining assays, the positive and negative controls showed the expected qualitative direct cell responses. Thereby, green fluorescence indicates living cells, while red fluorescence indicates dead cells. Good cell attachment onto the surface is displayed by spindle-shaped cell morphology, while a round morphology suggests a lack of adhesion. Overall, the HF-treated group showed similar results to those of the negative control, although the attachment of the MC3T3 even appeared to be improved (Figure 5). Untreated magnesium exhibited poor cell attachment with some red dead cells. None of the magnesium specimens showed gas bubble dissolution in the direct test.



**Figure 5.** Attachment of cells on surfaces of the different variants and controls. The pictures show the attachment, vitality and morphology of the cells. Green: vital cells; red: dead cells. Spindle-shaped morphology indicates healthy cells with firm attachment. Round cells indicate poor attachment onto the surface.

#### 2.3. Histopathological Results

All 18 test animals completed the study. No signs of infection or wound dehiscence could be observed.

At 6 weeks post implantation, the histological analyses showed that comparable amounts of newly formed bone were found within the former defect areas of all study groups (Figure 8D). At this study time point, only remnants of the magnesium meshes in the respective study groups were detectable (Figures 6 and 7). Moreover, no histological signs of the collagen membranes have been detected in these groups. Additionally, in the study group of the pure collagen membrane, no histological signs of

the biomaterial were visible. In the sham operation group, only within the central regions of the former defect areas a thin layer of dense vessel- and cell-poor connective tissue, including mostly fibroblasts and some single macrophages, were detected at this earliest study time point.



**Figure 6.** Histopathological comparison of both treated and HF-treated membranes. Images of Masson-Goldner (**C**,**D**) and Von Kossa (**A**,**B**,**E**,**F**) staining of the implantation site at 6, 12 and 18 weeks (100× magnifications, scalebars = 100  $\mu$ m). Left for untreated (Mg) and right for HF-treated (Mg<sup>#</sup>) magnesium. Mg is mainly degraded via dissolution and scarcely through phagocytic processes. Mg-HF however, is primarily being resorbed via active phagocytosis and non-cellular dissolution only plays a minor role. After degradation of the HF-coating, decomposition, as with untreated Mg, principally occurs non-cellular-driven but through dissolution. Yellow arrows = phagocytic cells, black arrows = fibroblasts, asterisks = slight fibrosis, white arrows = septa between the gas cavities.

Clear differences in the tissue responses to the HF-treated and the untreated magnesium materials were histologically observable (Figure 7). In the case of the untreated magnesium meshes, the mesh bars were surrounded by distinctly pronounced gas cavities. The magnesium bars were only connected to the surrounding connective tissue in small contact areas. Both the material- and the cavity-adherent connective tissue showed signs of a slight fibrosis involving mostly fibroblasts. Within the neighbored connective tissue, only moderate correlates of a material-associated inflammatory response, including mostly fibroblasts and low numbers of macrophages, have been detected at this study time point without signs of a phagocytic activity.



**Figure 7.** Gas cavity formation of HF- and untreated membranes.Representative images of Masson-Goldner (**A**,**F**) and Von Kossa (**B**–**E**) staining (40×) of the implantation site (scalebar =  $20 \mu m$ ) at 6, 12 and 18 weeks. Left for untreated (Mg) and right for HF-treated (Mg<sup>#</sup>) magnesium. Fibrotic capsule forming (\*) is visible at all times whereas gas cavity formation ceased to show after 12 weeks. The HF-coated mesh was mainly degraded by mononuclear cells (yellow arrows) up to 12 weeks, while the uncoated magnesium meshes elicited a fibrosis-like tissue reaction showing fibroblast accumulation (black arrows). Starting from 18 weeks after implantation, the tissue reaction in both groups was similar. Yellow arrows = phagocytic cells, black arrows = fibroblasts, asterisks = slight fibrosis.

In the group of the HF-treated magnesium meshes, histological signs of less pronounced gas cavities outgoing from the magnesium meshes could be detected (Figure 8A). Moreover, the tissue reaction to this material included a thin layer of phagocyting cells at the implant-tissue surfaces (Figure 7A,C,E). This cell layer was furthermore surrounded by a dense fibrotic capsule as also observed in the group of the untreated meshes (Figure 7B,D,E). Despite the observed different tissue responses and the magnesium-associated gas cavities, the magnesium meshes in both study groups were most of the in close contact to newly formed bone tissue (Figure 7).

1500000

1000000

500000

A)

area [µm²]





**Figure 8.** Overview of the various quantitative measurements. (A) Gas cavity surface in  $\mu m^2$  for uncoated (Mg; blue) and HF-treated (Mg-HF; red) magnesium after 6 and 12 weeks. HF-treated magnesium meshes show significantly lower gas cavity development compared to untreated magnesium meshes (\*\*: p < 0.01) up to 6 weeks after implantation. Gas cavity dimension of untreated magnesium meshes shows to be reduced significantly after 12 weeks (•: p < 0.05). (**B**–**D**) Bone regeneration as measured with contact radiography, DVT and in histomorphometry after 6, 12 and 18 weeks for uncoated (Mg; blue) and HF-coated magnesium (Mg-HF; red), collagen (Collagen; green) and control (Control; black). No significant differences (p > 0.05) were detectable.

At 8 weeks post implantation, still comparable amounts of newly formed bone tissue were observable in all four study groups and the amounts did visibly not differ from that observed at 6 weeks post implantation. At this study time point, still remnants of the magnesium meshes in the both respective study groups were detectable (Figures 6 and 7). The size of the material-related gas cavities in the group of the untreated magnesium meshes has visibly been decreased, while their peculiarity in the group of the HF-treated materials was comparably to the sizes at 6 weeks post implantation (Figure 8A). Moreover, the tissue responses were still comparable to that at 6 weeks post implantation (Figure 7). Thus, a fibrosis has been observed in case of the untreated magnesium meshes without any signs of phagocytic activities, while a thin cell layer of macrophages combined with a fibrosis-like capsule was found in the group of the HF-treated materials (Figure 7). No histological correlates of inflammatory processes were found in the surrounding connective tissue. In the group of the pure collagen membrane and the sham operation group, a thin layer of dense vessel- and cell-poor connective tissue including mostly fibroblasts were found only within the central regions of the former defect areas at this time point.

At 12 weeks post implantation, still comparable amounts of newly formed bone were found in all study groups and no differences that were additionally visibly comparable to the amounts at the former two study time points (Figure 8D). At this study time point, only very small remnants of the magnesium meshes could be microscopically detected (Figures 6 and 7). Interestingly, the tissue responses to the grids were equivalent at this latest study time point (Figure 7). In both groups, i.e., the group of the untreated magnesium meshes and the HF-treated material, no histological signs of gas cavities have been observed any longer and the remnants of both materials were surrounded by a thin cell layer, which was composed of single fibroblasts (Figure 7). No signs of a material-associated fibrosis or attached phagocytes could be observed at this time point. No signs of inflammatory processes were detected in the surrounding connective tissue. Furthermore, the aforementioned layers of dense connective tissue within the central defect regions were still detectable in the group of the pure collagen membrane and the sham operation group.

#### 2.4. Digital Volume Tomography (DVT) Analysis

Digital volume tomography (DVT) showed no statistically significant differences between the amounts of newly built bone in the four different study groups at all time points (Figure 8B–D and Figure 9). After 6 weeks, 76.17%  $\pm$  9.50% newly built bone was found in the group of the HF-treated magnesium mesh, which was comparable to the amounts of regenerated bone in the groups of the uncoated mesh (75.69%  $\pm$  9.23%), collagen group (77.86%  $\pm$  7.59%) and control group (74.25%  $\pm$  7.13%). At 12 weeks post implantation, the DVT analysis showed similar results, namely 72.17%  $\pm$  11.05% in the group of the HF-treated magnesium mesh and comparable 70.23%  $\pm$  7.49% in the groups of the uncoated mesh, the collagen group (77.11%  $\pm$  8.50%) and in the control group (70.98%  $\pm$  11.50%). At 18 weeks post implantation, equal results for the HF-treated magnesium mesh (75.50%  $\pm$  19.24%), the uncoated mesh (77.93%  $\pm$  9.94%), the collagen group (78.59%  $\pm$  13.88%) and the control group (78.58%  $\pm$  12.19%) were observed.



Figure 9. Imaging procedures to detect bone regeneration. (A) Three-dimensional DVT reconstruction and (B) radiological image as used for bone surface regeneration measurements. Selected circular defect area (C) before and (D) after application of color threshold.

#### 2.5. Contact Radiography

Overall, ossification increased over time in all study groups (Figures 8B–D and 9). Thereby, the extent of ossification was lower compared to DVT. After 6 weeks, bone regeneration of more than

60% was noticeable in all groups, while the empty control group tend to ossify at a slower extent. However, this effect is not significant. At all time points, no significant differences between all study groups (p > 0.05) could be detected.

After 6 weeks, an amount of  $62.37\% \pm 13.77\%$  new built bone was found in the group of the HF-treated magnesium mesh, which was comparable to the amounts of rebuilt bone in the groups of the uncoated mesh ( $66.78\% \pm 5.11\%$ ), the collagen group ( $67.18\% \pm 14.29\%$ ) and in the control group without biomaterial insertion ( $55.64\% \pm 12.98\%$ ).

At 12 weeks post implantation, the DVT analysis showed similar results, namely  $64.26\% \pm 16.42\%$  in the group of the HF-treated magnesium mesh and comparable  $69.45\% \pm 10.80\%$  in the groups of the uncoated mesh, the collagen group ( $68.59\% \pm 12.90\%$ ) and in the control group ( $59.50\% \pm 19.61\%$ ).

At 18 weeks post implantation, equal results for the HF-treated magnesium mesh (74.07%  $\pm$  15.28%), the uncoated Mg mesh (69.85%  $\pm$  5.61%), the collagen group (69.15%  $\pm$  10.84%) and the control group (70.49%  $\pm$  11.96%) were observed.

#### 2.6. Histomorphometrical Analysis

Connective tissue fillings as well as island-shaped ossifications in the defect area showed an increase over time for all groups. There was no indication of membrane collapse for neither of the membranes. The mean values of bone surface regeneration were clearly below those of DVT and contact radiography.

The quantitative measurements of bone regeneration using histomorphometrical analysis showed that no statistically significant differences between the amounts of newly regenerated bone were detectable within the four different study groups at 6 weeks post implantation (Figure 8D). Thus, an amount of  $46.48\% \pm 11.42\%$  new built bone was found in the group of the HF-treated magnesium mesh, which was comparable to the amounts of rebuild bone in the groups of the uncoated mesh (50.83%  $\pm$  22.09%), the collagen group (49.40%  $\pm$  17.55%) and the control group without biomaterial insertion (47.88%  $\pm$  22.34%).

At 12 weeks post implantation, the DVT analysis showed similar results, namely  $41.21\% \pm 29.17\%$  in the group of the HF-treated magnesium mesh and comparable  $52.02\% \pm 19.55\%$  in the groups of the uncoated mesh, the collagen group ( $46.59\% \pm 18.34\%$ ) and the control group ( $62.20\% \pm 20.86\%$ ).

At 18 weeks post implantation, equal results for the HF-treated magnesium mesh (56.59%  $\pm$  33.60%), the uncoated mesh (59.79%  $\pm$  18.14%), the collagen group (74.39%  $\pm$  14.40%) and the control group (58.15%  $\pm$  10.46%) were observed.

#### 2.7. Results of the Gas Cavity Measurements

The quantitative measurements of gas cavity dimension via digitized total scans showed statistically significant differences between the HF-treated and uncoated magnesium meshes at 6 weeks post implantation. The surface of  $535,453 \pm 323,998 \ \mu\text{m}^2$  for uncoated magnesium showed to be significantly higher than the gas cavity dimension of  $173,713 \pm 129,697 \ \mu\text{m}^2$  found in the group of HF-coated meshes (p < 0.01) (Figure 8A).

At 12 weeks post implantation, the gas cavity measurements revealed lower results for the HF-treated (445,522 ± 44,457  $\mu$ m<sup>2</sup>) compared to the uncoated magnesium meshes (131,299 ± 129,697  $\mu$ m<sup>2</sup>). The reduction in cavity size for untreated magnesium meshes was found to be significant different compared to 6 weeks post implantation (*p* < 0.05).

Both magnesium meshes were associated with gas cavities, while they gradually decreased over time. Interestingly, the observed gas cavities did not interfere with bone regeneration. Moreover, the histomorphometrical analysis revealed that the size of gas cavities was significantly lower in the group of the HF-treated magnesium meshes up to 6 weeks (p < 0.01) after implantation. Furthermore, the gas cavity dimension of the untreated magnesium meshes reduced significantly after 12 weeks (p < 0.05). After 18 weeks, no gas cavities could be detected.

#### 3. Discussion

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In dentistry and other medical disciplines, bioresorbable GBR/GTR-membranes provide a valid treatment option for non-weight-bearing (bone) regeneration procedures, while special indications, such as high-volume defects, may require volume-stable membranes [12,13]. So far, the available volume-stable membranes consist of non-resorbable materials and require surgical removal after therapy. In this context, magnesium has manifoldly been shown to constitute a favorable biomaterial for the development of stabilizing structures due to its satisfying biocompatibility, biodegradability and mechanical properties [29,44–46]. As a promising surface modification for the delay of premature degradation, and thus, hydrogen release, hydrofluoric acid (HF)-treatments and the resulting magnesium fluoride (MgF<sub>2</sub>)-layers have been previously investigated for a broader range of applications, varying from orthopedic to cardiovascular and oropharyngeal applications [47–49]. Thereby, a generally higher corrosion resistance could be detected [37,47–55]. Therefore, the combination of a magnesium mesh embedded in a native collagen membrane allows volume-stable interventions and complete biodegradation.

Thus, the purpose of this study was the in vitro and in vivo investigation of a newly developed volume-stable and weight-bearing GBR/GTR-membrane. The combination of a native collagen membrane with an HF-treated magnesium mesh to control hydrogen evolution and the overall degradation process represents a novel approach of this research objective. To our best knowledge, this was the first in vitro/in vivo assessment of a HF-treated magnesium mesh integrated in a native collagen membrane for GBR/GTR therapy. These may preferably be used in trauma surgery as well as oral surgery.

In vitro, both HF-treated and untreated magnesium variants were tested for their basic cytocompatibility to ensure a safe in vivo application. Thereby, the HF-treated membrane exhibited promising cytocompatibility in the extract and direct tests, while the untreated membrane was within the toxic range in all extract assay. The HF membrane was also toxic in the LDH assay. However, the difference with the negative control was not significant, which in turn indicates only a slight toxicity. After 72 h extraction, the untreated magnesium showed clear signs of corrosion, while the HF-treated magnesium showed no macroscopic changes of the surface. This result supports the observation that HF-layers protect magnesium from premature degradation and retard the precipitation of corrosion products. In the direct tests, both magnesium variants showed many living cells with better cell attachment for the HF-treated magnesium. Due to the good results in the direct test, the untreated membrane qualified for further in vivo tests. In concordance with other in vitro studies, the results also showed that HF-treatment of magnesium leads to improved cytocompatibility, cell attachment and corrosion resistance [37,48]. Overall, the fabricated prototype was elastic and adaptable to the defect, so that volume stability can be insured. Thus, the new membrane exhibits similar properties as the already available non-absorbable PTFE membranes with integrated titanium meshes. However, further studies should investigate the applicability under real treatment conditions.

In vivo, 18 New Zealand White Rabbits were randomly divided into four study groups and three time points (6, 12, 18 weeks) as follows: (a) HF-treated magnesium mesh embedded in a native collagen membrane, (b) untreated magnesium mesh embedded in a native collagen membrane, (c) the native collagen membrane alone and (d) an empty control. The results of the study showed that total bone regeneration was similar for all groups at all time points, which questions the general necessity for GBR/GTR membranes. In this context, other studies discussed and showed similar results for uncovered bony defects compared to collagen membrane covered defects [56–59]. Moreover, the used animal model might be a reason for the comparable values as it does not reflect the clinical situation that is associated with the application of such a biomaterial. In this context, Byun et al. evaluated a hydroxyapatite (HA)-coated magnesium mesh in a rat calvarial model in a comparable experimental setup [60]. In terms of new bone formation, also no differences between all groups were found in the work of Byun et al. [60]. Regarding the microscopic appearance of the newly formed tissue, HA-coated meshes resulted in well-organized defect areas, while the empty controls seemed to be

curved, distorted and filled with malformed collagenous tissue. However, such qualitative differences were not visible in our study, which may be due to the use of another animal model or the use of collagen for embedding [60]. Altogether, these qualitative results of the present study are subjected to limitations, as the application of the new membrane and the control groups should be analyzed in a load-bearing defect model in combination with and without bone substitute materials. Thus, an animal model should be applied, which resembles human conditions.

Moreover, the values for bone regeneration were different for the applied measurement methods according to the order of DVT > contact radiography > histomorphometry. These results may be explained by the different resolution capacities of the applied imaging procedures. Additionally, it has to be mentioned that both DVT and contact radiography allow for analysis of the complete defect sides, while the histomorphometrical method only enables for analysis of a representative 2D slide from the defect area. Since magnesium and bone have similar densities, different distinction properties of the applied measurements methods between bone tissue and the magnesium implants might be a further reason for the different results [61].

The histopathological observations in combination with the histomorphometrical analysis of the magnesium-associated gas cavity evolution are the most interesting results of the present study, even in view of the applicability of HF-treatment for the adaption of the degradation behavior of magnesium implants. The histopathological analysis showed that the HF-induced surface coating prevented gas cavity development up to 12 weeks compared to the values found in the group of the untreated magnesium implants, while these effects were no longer detectable after 18 weeks. Similar observations on delayed gas release of MgF<sub>2</sub>-treated materials could be shown in other studies as well [53–55]. Thereby, gas bubble release delayed between 90 and 180 days [53–55]. Compared with other coating technologies, HF-treatment can partially prevent a premature burst of major gas bubble formation [30,62]. However, the applicability for larger bone defects or fracture plates that require mechanical strength for about 3 months must be critically discussed and tested [63].

Moreover, the histopathological analysis revealed that the HF-treated magnesium meshes induced a layer of mononuclear phagocytes up to 12 weeks, in combination with a slight fibrosis involving mostly fibroblasts, while the untreated magnesium meshes only induced a fibrosis-like tissue reaction. After 18 weeks, the tissue reaction in both groups has shown to be similar. These result initially lead to the conclusion that the surface treatment that has already been shown to lead to the formation of a magnesium fluoride MgF<sub>2</sub>-layer onto the surface of such kind of implants has been phagocytosed within 12 weeks after implantation but enables to delay the hydrogen release and thus the gas cavity development.

In this context, in the work of Byun et al., no hydrogen gas release could be detected, most likely explained by the different coating or by rapid encapsulation and integration of the HA-mesh into the surrounding tissue [60]. In both studies, the implantation of the magnesium membranes triggered an inflammatory tissue reaction, being described as exaggerated by Byun et al. after 18 weeks [60]. Other studies described mild or unspecific inflammatory responses to HF-treated magnesium [53–55,64]. In the present study, both the HF-treated and the untreated magnesium meshes only showed minor inflammation, which decreased over time and seems to be driven by mononuclear phagocytosis for the MgF<sub>2</sub> layer. Unfortunately, the cell composition involved in decomposition of the mesh was not further described by Byun et al., or other studies, and no comparable studies have been published until now dealing with this scientific topic [53–55,60,64].

Furthermore, the present observations lead to the conclusion that the hydrogen gas cavities are covered with fibrotic tissue. In this context, it has already been discussed that a biomaterial-related fibrosis might be a restricting factor for its regenerative capacities due to its isolation from the surrounding tissue [65–69]. However, other study results refute this assumption as it has been shown in the case of bone substitute materials, which partially also induce slight fibrotic encapsulation, that this tissue reaction does not interfere with the biomaterial-associated healing processes [70–72]. The latter is evidenced by the observation in the present study that—despite the observed different

tissue responses—the magnesium meshes in both study groups were in close contact to newly formed bone tissue and, thus, did not interfere with the formation of new bone tissue.

This is the first study that provides results of magnesium-based meshes embedded in a collagen membrane for volume stable GBR/GTR therapy. Both in vitro and in vivo results are congruent with respect to improved cyto- and biocompatibility of HF-treated membranes. Nevertheless, this study and its results are subjected to limitations and further questions appeared. In vitro, corrosion measurements by potentiodynamic polarization or in a corrosion testing device can improve the results and the further applicability of the tested specimens. Further methods, such as hemocompatibility or tests for osteogenic differentiation, can be applied, as displayed in other studies of our group [29]. Furthermore, the mechanical characteristics of the MgF<sub>2</sub> layer should additionally be tested before and during its application. In vivo, magnesium showed not to be fully degraded, which enquires the extension of the study period. A more detailed comparison of radiographic methods for measuring bone regeneration could provide further approaches to reduce in vivo studies. Closer examination of inflammatory relationships (e.g., the involvement of different cell types and the alignment of macrophages) could also be of interest for further development of this material class [19]. Finally, the direct comparison or amplification of other surface modifications such as duplex- (MgF<sub>2</sub>/PCL) [50], nanocomposite- [73] and composite-coatings [51,74] as well as the extension of other magnesium alloys [47–51,75–81] should be investigated and might promise new results in the future. Overall, especially in comparison with other studies, there is further need for simplification and standardization of in vitro and in vivo studies of magnesium-based materials.

#### 4. Materials and Methods

#### 4.1. Biomaterial Preparation

Magnesium sheets made of magnesium alloy AZ31 (biotrics bioimplants GmbH, Berlin, Germany) with dimensions of  $30 \times 40$  mm were subjected for further processing. The samples were ultrasonically cleaned in 100% ethanol and distilled water. MgF<sub>2</sub> coating was achieved by immersion of the test samples in 20% HF solution at 37 °C for 6 h and subsequently cleaned in 100% ethanol and distilled water. For the in vitro assessment, untreated and HF-treated sheets were used in dimensions of  $0.5 \times 0.5$  cm<sup>2</sup>. For the in vivo assessment, untreated magnesium sheets were further processed using femtosecond laser cut to achieve a diamond pattern before treatment of the meshes by means of HF. Both treated and untreated magnesium meshes were embedded in a native porcine collagen membrane (Jason<sup>®</sup> Membran, botiss biomaterials GmbH Berlin, Germany) that was further used as the negative control in vivo. Sterilization was achieved by gamma irradiation.

#### 4.2. Cytocompatibility Analysis

Cytocompatibility analyses were executed in accordance with the ISO 10993-5/-12 and were already described in detail previously [29,40,63]. The experimental setup is described briefly in the following paragraphs.

#### 4.2.1. Reference Material

All reference materials were sterilized by immersion in isopropanol for 5 min with subsequent drying in a laminar flow hood. RM-A (Hatano Research Institute, Food and Drug Safety Center, Ochiai, Japan) was used as a positive control reference. Wako plastic sheets (Wako Pure Chemical Industries, Ltd., Osaka, Japan, Cat. No.160-08893) were used as a nontoxic control material. Samples of RM-A, Wako plastic sheets and titanium were prepared with the same surface areas as the material specimens and sterilized likewise.

#### 4.2.2. Cells and Cell Culture

Mouse fibroblasts of the L-929 cell line and mouse osteoblast precursor cells of the MC3T3 cell line were obtained from the European Collection of Cell Cultures, ECACC (Salisbury, UK). Cells were cultured in cell culture medium (MEM (Minimum Essential Medium)) supplemented with 10% fetal bovine serum, penicillin/streptomycin (100 U/mL each) (all from Life Technologies, Carlsbad, CA, USA) and L-glutamine (Sigma-Aldrich, St. Louis, MO, USA) to a final concentration of 4 mM under cell culture conditions (37 °C, 5% CO<sub>2</sub>, and 95% humidity). At about 80% confluency, cells were passaged.

#### 4.2.3. Extract Analysis

#### Extraction

Test and control samples were extracted for 72 h at a surface to volume ratio of 3 cm<sup>2</sup>/mL in cell culture medium under cell culture conditions. Cell culture medium alone was incubated under identical conditions to serve as a negative control extract. After removal of the specimens, the remaining extracts were centrifuged at 14,000 rpm for 10 min. The supernatants were used for the different assays that are described below.

#### Assay Procedure

In total, 96 well plates were seeded with  $1 \times 10^4$  L-929 cells/well in 100 µL cell culture medium and incubated under cell culture conditions for 24 h. Thereafter, cell culture medium was discarded and 100 µL of extract were added to each well. Cells were further incubated for 24 h and then subjected to the BrdU- and XTT-assays while the supernatants were subjected to the Lactate Dehydrogenase (LDH)-assay. Identical assays but omitting cells were conducted for all extracts as a control for assay interference. Blank controls (medium alone without cells) were subtracted from the absorbance values in all assays.

#### Bromodeoxyuridine/5-Bromo-2'-Deoxyuridine (BrdU)-Assay

A BrdU (colorimetric) test kit (Roche Diagnostics, Mannheim, Germany) was used according to the manufacturer's instructions. Briefly, cells were labeled with BrdU for 2 h under cell culture conditions and subsequently fixed for 30 min at room temperature with a FixDenat reagent. Then, the fixed cells were incubated for 1 h with an anti-BrdU-peroxidase (POD) antibody and washed 3 times for 5 min with washing buffer. The immune complexes were detected after a subsequent substrate reaction with tetramethyl-benzidine (TMB) (20 min at room temperature) followed by the addition of  $25 \ \mu L 1 M H_2 SO_4$  to stop the reaction using a scanning multi-well spectrophotometer (ELISA reader) with filters for 450 and 690 nm (reference wavelength).

# Sodium 3,3'-[1(Phenylamino)carbonyl]-3,4-tetrazolium]-3is(4-methoxy-6-nitro) Benzene Sulfonic Acid Hydrate (XTT)-Assay

The Cell Proliferation Kit II (Roche Diagnostics, Mannheim, Germany) was used according to the manufacturer's instructions. Briefly, the electron-coupling reagent was mixed with the XTT labeling reagent (1:50 dilution) and 50  $\mu$ L of the mixture was added to the cells. After 4 h of incubation under cell culture conditions, substrate conversion was quantified by measuring the absorbance of 100  $\mu$ L aliquots in a new 96 well plate using a scanning multi-well spectrophotometer (ELISA reader) with filters for 450 and 650 nm (reference wavelength).

#### Lactate Dehydrogenase (LDH)-Assay

The LDH-Cytotoxicity Assay Kit II (BioVision, Milpitas, CA, USA) was used according to the manufacturer's instructions. Briefly, 10  $\mu$ L of the cell supernatants were incubated with 100  $\mu$ L LDH reaction mix for 30 min at room temperature. After the addition of stopping solution, absorbances

were measured using a scanning multi-well spectrophotometer (ELISA reader) with filters for 450 and 650 nm (reference wavelength).

#### 4.2.4. Live-Dead Staining

Mg-test samples and controls were seeded with  $2.4 \times 10^5$  L-929-cells and  $2.0 \times 10^5$  MC3T3-cells in 1 mL medium in each well of 12 well plates (the surface area/medium ratio was 5.65 cm<sup>2</sup>/mL). Assays were carried out after 24 h incubation under cell culture conditions. In order to perform live-dead cell staining on the surfaces of the specimens, 60 µL per mL medium propidium iodide (PI) stock solution (50 µg/mL in PBS) and 500 µL per mL medium fresh fluorescein diacetate (FDA) working solution (20 µg/mL in PBS from 5 mg/mL FDA in acetone stock solution) were added to each well (12 well plate). After a brief incubation for 3 min at room temperature, specimens were rinsed in prewarmed PBS and were immediately examined with an upright fluorescence microscope (Nikon ECLIPSE Ti-S/L100, Nikon GmbH, Düsseldorf, Germany) equipped with a filter for parallel detection of red and green fluorescence. Pictures were taken using a 4×, 10× and 20× objective. Cells visualized using the 10× objective were counted using the software ImageJ. For each material, cells were counted from three experiments to calculate mean and standard deviation.

#### 4.3. Experimental Animals and Surgical Procedure

Eighteen female New Zealand White Rabbits (Charles River Laboratories, Wilmington, MA, USA), randomly divided into three groups (6, 12, 18 weeks) weighing between 2.2–2.9 kg at the age of approximately 12 weeks, were used for the experiments. The rabbits were acclimatized for four weeks upon delivery, had ad libitum access to water and dry food, and were kept under a constant day-night cycle. This study was conducted at the Animal Research Laboratory of the University Medical Center Hamburg-Eppendorf by the Department of Oral and Maxillofacial Surgery and was approved by the Veterinary Office of the Hamburg Authority for Health and Consumer Protection (No. 29/16, approval date: 19 May 2016).

Prior to implantation, the animals were anaesthetized with 5 mg/kg xylazine (Rompun<sup>®</sup> 2%, Bayer Vital GmbH, Leverkusen, Germany) and 70 mg/kg ketamine (Ketamin 10%, bela-pharm GmbH & Co. KG, Vechta, Germany). 11 animals additionally received buprenorphine (Buprenovet<sup>®</sup>, 0.3 mg/mL, Bayer Vital GmbH, Leverkusen, Germany) and 7 animals received Fentanyl (5–20 µg/kg/h) as an opioid. The calvariae of the rabbits were shaved and the skin was disinfected with Cutasept<sup>®</sup> F (BODE Chemie GmbH, Hamburg, Germany). A transverse skin incision was made along the midline of the skull, and the periosteum was elevated laterally. Four circular defects, of each 8 mm in diameter were created (Figure 10) (Meisinger, Neuss, Germany, Ref. No. 330205486001070). After ensuring adequate hemostasis using electrocoagulation, each defect was either covered by a porcine collagen membrane (Jason<sup>®</sup> Membran, botiss biomaterials GmbH, Berlin, Germany), a magnesium reinforced collagen membrane with a randomized trial protocol. One defect remained empty and served as control.

After this insertion procedure, the periosteum layer and the muscle were relocalized and stitched with individual button sutures (4-0 Vicryl<sup>®</sup>, Ethicon Inc., Somerville, NJ, USA). Finally, the skin was closed with surgical staples and the surgical site was treated with a wound spray (Desitin<sup>®</sup> Salbenspray, Desitin Arzneimittel GmbH, Hamburg, Germany). Each animal received 10 mg/kg enrofloxacin (Baytril<sup>®</sup> 10%, Bayer Vital GmbH, Leverkusen, Germany) as perioperative antibiotic prophylaxis up to 5 days post-surgery.

Six rabbits each were sacrificed at each study time point, i.e., after 6, 12 and 18 weeks, by intracardial injection of 3–5 mL T61 (T61, MSD Tiergesundheit, Luzern, Switzerland) after general anesthesia by the abovementioned anesthetics. Immediately, the calvaria, liver, kidneys, and spleens and the calvarium containing the implants were harvested and immerged in a solution of 4% formaldehyde for 48 h.



**Figure 10.** Intraoperative view on calvarial implantation model. Creation of four circular defects using a trephine burr (**A**,**B**), followed by placement of membranes (**C**). Image of the hydrofluoric acid (HF)-treated magnesium (Mg) mesh in a native collagen membrane (**D**).

#### 4.4. Digital Volume Tomography (DVT) Analysis

Digital volume tomography (DVT) was carried out by a ProMax unit (Planmeca, Finland) using a voltage of 90 kV and an X-ray tube current of 10 mA. A three-dimensional reconstruction of the calvarial defects was created by the open source program "Horos" (https://www.horosproject.org) using a slice thickness 6.45 mm. The bone surface measurement was then performed by the open source program "ImageJ" (https://imagej.nih.gov/ij/) and was limited to a circular area of 8 mm in diameter, thus, the size of the defect created. The color threshold of bone density was set for each defect individually within the range of 73 to 137 using a color depth of 8-bit.

#### 4.5. Radiological Analysis

Contact radiography was performed by an LX-60 (Faxitron X-ray LLC, Tucson, AZ, USA) with exposure parameters of 1000 ms at 26 kV. The ImageJ software was then used to perform bone surface measurements on the resulting TIF-file analogous to the DVT analysis.

#### 4.6. Histological Work Up

After completion of DVT and radiological imaging, each defect was separated sagittally, resulting in two halves. Subsequently, dehydration via a series of increasing alcohol concentrations and a final exposure to methyl methacrylate (MMA), benzoyl peroxid (BPO) and nonylphenyl-polyethyleneglycol acetate were performed before paraffin embedding. Cross sections with a thickness of 35  $\mu$ m were prepared by means of a Leica SM 2500E (Leica Microsystems GmbH, Wetzlar, Germany) and stained with Masson's trichrome stain, Kossa van Gieson's stain and Toluidine blue stain. Undecalcified thin section preparation (30  $\mu$ m) was performed on the remaining half [82,83].

#### 4.7. Histopathological Analysis

The histopathological analysis focused on the outcome of the material-mediated bone defect healing and the comparison of the tissue reactions to the different membrane materials based on a previously published protocol [19,41–43]. A light microscope Axio Scope.A1 (Carl Zeiss Microscopy

GmbH, Jena, Germany) was used for the analysis. These analyses included the evaluation of the following parameters: fibrosis, hemorrhage, necrosis, vascularization and the presence of neutrophils, lymphocytes, plasma cells, macrophages and multinucleated giant cells (BMGCs). Microphotographs were taken with the cellSens Entry (Olympus K.K., Tokyo, Japan) software using a DP72 (Olympus K.K., Tokyo, Japan) digital camera connected to the microscope.

## 4.8. Histomorphometrical Analysis

The histomorphometrical analyses included the comparative measurements of the amount of regenerated bone, which was measured under 20× magnification using a Zeiss 47 52 69 light microscope (Carl Zeiss Microscopy GmbH, Jena, Germany) and a calibrated scale. Additionally, the size of cavities related to the magnesium grids were histomorphometrically measured. Briefly, so-called "total scans" were generated with the aid of a specialized scanning microscope, which consists of an Axio Scope.A1 (Carl Zeiss Microscopy GmbH, Jena, Germany) combined with a digital camera and an automatic scanning table (Märzhäuser, Wetzlar, Germany) connected to an PC system running the Zen Core software (Zeiss, Tokyo, Japan). The resulting images were composed of 100 to 120 single images with a 100× magnification in a resolution of 2500 × 1200 pixels and contained the complete implant area as well as the peri-implant tissue. For the histomorphometrical measurement of the cavities, the slides stained by Toluidine blue were digitized. These images allowed the measurements of the cavity sizes by means of the Zen Core software.

#### 4.9. Statistical Analysis

Quantitative data were statistically analyzed via an analysis of variance (ANOVA) and a following Bonferroni post-hoc test via the SPSS software (SPSS 24, IBM, Armonk, NY, USA). Statistical differences were designated as significant if *p*-values were less than 0.05 (\*  $p \le 0.05$ ) and highly significant if *p*-values were less than 0.01 (\*\*  $p \le 0.01$ ). Finally, the data were shown as mean ± standard deviation.

## 5. Conclusions

HF-treated magnesium meshes embedded in native collagen-based barrier membranes represent a promising volume stable alternative for GBR/GTR therapy in comparison to conventional non-resorbable membranes. The material fulfills the requirements for cyto- and biocompatibility, but should be more intensively be investigated in further studies.

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#### References

- Basler, T.; Naenni, N.; Schneider, D.; Hämmerle, C.H.; Jung, R.E.; Thoma, D.S. Randomized controlled clinical study assessing two membranes for guided bone regeneration of peri-implant bone defects: 3-year results. *Clin. Oral Implant. Res.* 2018, 29, 499–507. [CrossRef] [PubMed]
- Milinkovic, I.; Cordaro, L. Are there specific indications for the different alveolar bone augmentation procedures for implant placement? A systematic review. *Int. J. Oral Maxillofac. Surg.* 2014, 43, 606–625. [CrossRef] [PubMed]
- Smeets, R.; Jung, O.; Hanken, H.; Hartjen, P.; Al Dam, A.; Gröbe, A.; Heiland, M.; Gosau, M.; Rothamel, D.; Schlee, M. Was können regenerative Materialien in der Zahnmedizin leisten–und wo sind die Grenzen. Deutsceh Zahnärztliceh Zeitschrift. 2014, 69, 708–721.
- Linde, A.; Alberius, P.; Dahlin, C.; Bjurstam, K.; Sundin, Y. Osteopromotion: A soft-tissue exclusion principle using a membrane for bone healing and bone neogenesis. *J. Periodontol.* 1993, 64, 1116–1128. [CrossRef] [PubMed]
- 5. Linde, A.; Thorén, C.; Dahlin, C.; Sandberg, E. Creation of new bone by an osteopromotive membrane technique: An experimental study in rats. *J. Oral Maxillofac. Surg.* **1993**, *51*, 892–897. [CrossRef]
- 6. Karring, T.; Nyman, S.; Gottlow, J.; Laurell, L. Development of the biological concept of guided tissue regeneration—Animal and human studies. *Periodontology* 2000 **1993**, *1*, 26–35. [CrossRef]
- 7. Meinig, R.P. Clinical use of resorbable polymeric membranes in the treatment of bone defects. *Orthop. Clin. N. Am.* **2010**, *41*, 39–47. [CrossRef]
- Dimitriou, R.; Mataliotakis, G.I.; Calori, G.M.; Giannoudis, P.V. The role of barrier membranes for guided bone regeneration and restoration of large bone defects: Current experimental and clinical evidence. *BMC Med.* 2012, *10*, 81. [CrossRef]
- Gottlow, J. Guided Tissue Regeneration using bioresorbable and nonresorbable devices. J. Jpn. Soc. Periodontol. 1993, 35, 37. [CrossRef]
- Scantlebury, T.V. 1982-1992: A decade of technology development for guided tissue regeneration. *J. Periodontol.* 1993, 64, 1129–1137. [CrossRef]
- 11. Ghanaati, S. Non-cross-linked porcine-based collagen I-III membranes do not require high vascularization rates for their integration within the implantation bed: A paradigm shift. *Acta Biomater.* **2012**, *8*, 3061–3072. [CrossRef] [PubMed]
- Sheikh, Z.; Hamdan, N.; Ikeda, Y.; Grynpas, M.; Ganss, B.; Glogauer, M. Natural graft tissues and synthetic biomaterials for periodontal and alveolar bone reconstructive applications: A review. *Biomater. Res.* 2017, 21, 9. [CrossRef] [PubMed]
- Tolstunov, L.; Hamrick, J.F.E.; Broumand, V.; Shilo, D.; Rachmiel, A. Bone Augmentation Techniques for Horizontal and Vertical Alveolar Ridge Deficiency in Oral Implantology. *Oral Maxillofac. Surg. Clin. N. Am.* 2019, *31*, 163–191. [CrossRef] [PubMed]
- 14. Smeets, R.; El-Moawen, A.; Jung, O.; Hanken, H.; Hartjen, P.; Heiland, M.; Kansy, K.; Kloss, F.; Kolk, A. From bench to application: Current practices in tissue engineering and its realisation at maxillofacial units in Germany, Austria and Switzerland. *J. Craniomaxillofac. Surg.* **2014**, *42*, 1128–1132. [CrossRef]
- 15. Lundgren, A.; Sennerby, L.; Lundgren, D. Guided jaw-bone regeneration using an experimental rabbit model. *Int. J. Oral Maxillofac. Surg.* **1998**, *27*, 135–140. [CrossRef]
- 16. Wiltfang, J.; Merten, H.-A.; Peters, J.-H. Comparative study of guided bone regeneration using absorbable and permanent barrier membranes: A histologic report. *Int. J. Oral Maxillofac. Implant.* **1998**, *13*, 416–421.
- 17. Liu, J.; Kerns, D.G. Mechanisms of guided bone regeneration: A review. *Open Dent. J.* **2014**, *8*, 56–65. [CrossRef]
- 18. Elgali, I.; Omar, O.; Dahlin, C.; Thomsen, P. Guided bone regeneration: Materials and biological mechanisms revisited. *Eur. J. Oral Sci.* **2017**, *125*, 315–337. [CrossRef]
- Korzinskas, T.; Jung, O.; Smeets, R.; Stojanovic, S.; Najman, S.; Glenske, K.; Hahn, M.; Wenisch, S.; Schnettler, R.; Barbeck, M. In Vivo Analysis of the Biocompatibility and Macrophage Response of a Non-Resorbable PTFE Membrane for Guided Bone Regeneration. *Int. J. Mol. Sci.* 2018, *19*, 2952. [CrossRef]
- 20. Sanctis, M.; Zucchelli, G.; Clauser, C. Bacterial colonization of barrier material and periodontal regeneration. *J. Clin. Periodontol.* **1996**, *23*, 1039–1046. [CrossRef]

- 21. Tempro, P.J.; Nalbandian, J. Colonization of retrieved polytetrafluoroethylene membranes: Morphological and microbiological observations. *J. Periodontol.* **1993**, *64*, 162–168. [CrossRef] [PubMed]
- 22. Selvig, K.A.; Kersten, B.G.; Chamberlain, A.D.H.; Wikesjö, U.M.; Nilvúus, R.E. Regenerative surgery of intrabony periodontal defects using ePTFE barrier membranes: Scanning electron microscopic evaluation of retrieved membranes versus clinical healing. *J. Periodontol.* **1992**, *63*, 974–978. [CrossRef] [PubMed]
- 23. Akbay, E.; Aydogan, F. Reconstruction of isolated mandibular bone defects with non-vascularized corticocancellous bone autograft and graft viability. *Auris Nasus Larynx* **2014**, *41*, 56–62. [CrossRef] [PubMed]
- 24. Zheng, Y.; Gu, X.; Witte, F. Biodegradable metals. Mater. Sci. Eng. R Rep. 2014, 77, 1–34. [CrossRef]
- 25. Staiger, M.P.; Pietak, A.M.; Huadmai, J.; Dias, G. Magnesium and its alloys as orthopedic biomaterials: A review. *Biomaterials* **2006**, *27*, 1728–1734. [CrossRef]
- Zhao, D.; Witte, F.; Lu, F.; Wang, J.; Li, J.; Qin, L. Current status on clinical applications of magnesium-based orthopaedic implants: A review from clinical translational perspective. *Biomaterials* 2017, 112, 287–302. [CrossRef]
- 27. Witte, F.; Fischer, J.; Nellesen, J.; Crostack, H.A.; Kaese, V.; Pisch, A.; Beckmann, F.; Windhagen, H. In vitro and in vivo corrosion measurements of magnesium alloys. *Biomaterials* **2006**, *27*, 1013–1018. [CrossRef]
- Witte, F.; Kaese, V.; Haferkamp, H.; Switzer, E.; Meyer-Lindenberg, A.; Wirth, C.J.; Windhagen, H. In vivo corrosion of four magnesium alloys and the associated bone response. *Biomaterials* 2005, 26, 3557–3563. [CrossRef]
- Jung, O.; Smeets, R.; Hartjen, P.; Schnettler, R.; Feyerabend, F.; Klein, M.; Wegner, N.; Walther, F.; Stangier, D.; Henningsen, A.; et al. Improved In Vitro Test Procedure for Full Assessment of the Cytocompatibility of Degradable Magnesium Based on ISO 10993-5/-12. *Int. J. Mol. Sci.* 2019, 20, 255. [CrossRef]
- 30. Hornberger, H.; Virtanen, S.; Boccaccini, A. Biomedical coatings on magnesium alloys—A review. *Acta Biomater.* **2012**, *8*, 2442–2455. [CrossRef]
- Wang, J.; Tang, J.; Zhang, P.; Li, Y.; Wang, J.; Lai, Y.; Qin, L. Surface modification of magnesium alloys developed for bioabsorbable orthopedic implants: A general review. *J. Biomed. Mater. Res. B Appl. Biomater.* 2012, 100, 1691–1701. [CrossRef] [PubMed]
- 32. Tian, P.; Liu, X. Surface modification of biodegradable magnesium and its alloys for biomedical applications. *Regen. Biomater.* **2015**, *2*, 135–151. [CrossRef] [PubMed]
- 33. Thomann, M.; Krause, C.; Angrisani, N.; Bormann, D.; Hassel, T.; Windhagen, H.; Meyer-Lindenberg, A. Influence of a magnesium-fluoride coating of magnesium-based implants (MgCa0. 8) on degradation in a rabbit model. *J. Biomed. Mater. Res. A* **2010**, *93*, 1609–1619. [PubMed]
- 34. Carboneras, M.; García-Alonso, M.; Escudero, M. Biodegradation kinetics of modified magnesium-based materials in cell culture medium. *Corros. Sci.* **2011**, *53*, 1433–1439. [CrossRef]
- 35. Yan, T.; Tan, L.; Xiong, D.; Liu, X.; Zhang, B.; Yang, K. Fluoride treatment and in vitro corrosion behavior of an AZ31B magnesium alloy. *Mater. Sci. Eng. C* 2010, *30*, 740–748. [CrossRef]
- 36. Witte, F.; Fischer, J.; Nellesen, J.; Vogt, C.; Vogt, J.; Donath, T.; Beckmann, F. In vivo corrosion and corrosion protection of magnesium alloy LAE442. *Acta Biomater.* **2010**, *6*, 1792–1799. [CrossRef]
- Kang, M.H.; Jang, T.S.; Kim, S.W.; Park, H.S.; Song, J.; Kim, H.E.; Jung, K.H.; Jung, H.D. MgF2-coated porous magnesium/alumina scaffolds with improved strength, corrosion resistance, and biological performance for biomedical applications. *Mater. Sci. Eng. C Mater. Biol. Appl.* 2016, 62, 634–642. [CrossRef]
- Da Conceicao, T.; Scharnagl, N.; Blawert, C.; Dietzel, W.; Kainer, K. Surface modification of magnesium alloy AZ31 by hydrofluoric acid treatment and its effect on the corrosion behaviour. *Thin Solid Films* 2010, *518*, 5209–5218. [CrossRef]
- 39. Sieger, D.; Korzinskas, T.; Jung, O.; Stojanovic, S.; Wenisch, S.; Smeets, R.; Gosau, M.; Schnettler, R.; Najman, S.; Barbeck, M. The Addition of High Doses of Hyaluronic Acid to a Biphasic Bone Substitute Decreases the Proinflammatory Tissue Response. *Int. J. Mol. Sci.* **2019**, *20*, 1969. [CrossRef]
- 40. Jung, O.; Smeets, R.; Porchetta, D.; Kopp, A.; Ptock, C.; Muller, U.; Heiland, M.; Schwade, M.; Behr, B.; Kroger, N.; et al. Optimized in vitro procedure for assessing the cytocompatibility of magnesium-based biomaterials. *Acta Biomater.* **2015**, *23*, 354–363. [CrossRef]

- Tawil, G.; Barbeck, M.; Unger, R.; Tawil, P.; Witte, F. Sinus Floor Elevation Using the Lateral Approach and Window Repositioning and a Xenogeneic Bone Substitute as a Grafting Material: A Histologic, Histomorphometric, and Radiographic Analysis. *Int. J. Oral Maxillofac. Implant.* 2018, 33, 1089–1096. [CrossRef] [PubMed]
- 42. Barbeck, M.; Booms, P.; Unger, R.; Hoffmann, V.; Sader, R.; Kirkpatrick, C.J.; Ghanaati, S. Multinucleated giant cells in the implant bed of bone substitutes are foreign body giant cells-New insights into the material-mediated healing process. *J. Biomed. Mater. Res. A* **2017**, *105*, 1105–1111. [CrossRef] [PubMed]
- 43. Barbeck, M.; Unger, R.E.; Booms, P.; Dohle, E.; Sader, R.A.; Kirkpatrick, C.J.; Ghanaati, S. Monocyte preseeding leads to an increased implant bed vascularization of biphasic calcium phosphate bone substitutes via vessel maturation. *J. Biomed. Mater. Res. A* **2016**, *104*, 2928–2935. [CrossRef] [PubMed]
- Chaya, A.; Yoshizawa, S.; Verdelis, K.; Myers, N.; Costello, B.J.; Chou, D.T.; Pal, S.; Maiti, S.; Kumta, P.N.; Sfeir, C. In vivo study of magnesium plate and screw degradation and bone fracture healing. *Acta Biomater* 2015, 18, 262–269. [CrossRef]
- 45. Chaya, A.; Yoshizawa, S.; Verdelis, K.; Noorani, S.; Costello, B.J.; Sfeir, C. Fracture healing using degradable magnesium fixation plates and screws. *J. Oral Maxillofac. Surg.* **2015**, *73*, 295–305. [CrossRef]
- 46. Ding, W. Opportunities and challenges for the biodegradable magnesium alloys as next-generation biomaterials. *Regen. Biomater.* **2016**, *3*, 79–86. [CrossRef]
- 47. Yu, W.; Zhao, H.; Ding, Z.; Zhang, Z.; Sun, B.; Shen, J.; Chen, S.; Zhang, B.; Yang, K.; Liu, M.; et al. In vitro and in vivo evaluation of MgF2 coated AZ31 magnesium alloy porous scaffolds for bone regeneration. *Colloids Surf. B Biointerfaces* **2017**, *149*, 330–340. [CrossRef]
- Mao, L.; Yuan, G.; Niu, J.; Zong, Y.; Ding, W. In vitro degradation behavior and biocompatibility of Mg-Nd-Zn-Zr alloy by hydrofluoric acid treatment. *Mater Sci. Eng. C Mater Biol. Appl.* 2013, 33, 242–250. [CrossRef]
- 49. Zhao, N.; Workman, B.; Zhu, D. Endothelialization of novel magnesium-rare earth alloys with fluoride and collagen coating. *Int. J. Mol. Sci.* **2014**, *15*, 5263–5276. [CrossRef]
- 50. Makkar, P.; Kang, H.J.; Padalhin, A.R.; Park, I.; Moon, B.G.; Lee, B.T. Development and properties of duplex MgF2/PCL coatings on biodegradable magnesium alloy for biomedical applications. *PLoS ONE* **2018**, *13*, e0193927. [CrossRef]
- Liu, W.; Yan, Z.; Ma, X.; Geng, T.; Wu, H.; Li, Z.J.M. Mg-MOF-74/MgF2 Composite Coating for Improving the Properties of Magnesium Alloy Implants: Hydrophilicity and Corrosion Resistance. *Materials* 2018, *11*, 396. [CrossRef] [PubMed]
- 52. Zhang, J.; Kong, N.; Niu, J.; Shi, Y.; Li, H.; Zhou, Y.; Yuan, G. Influence of fluoride treatment on surface properties, biodegradation and cytocompatibility of Mg–Nd–Zn–Zr alloy. *J. Mater. Sci. Mater. Med.* **2014**, 25, 791–799. [CrossRef] [PubMed]
- 53. Seitz, J.M.; Eifler, R.; Weber, C.; Lenarz, T.H.; Maier, H.J.; Durisin, M. In vivo degradation effects of alloy MgNd2 in contact with mucous tissue. *J. Biomed. Mater. Res. A* **2015**, *103*, 2427–2440. [CrossRef] [PubMed]
- Weber, C.M.; Eifler, R.; Seitz, J.M.; Maier, H.J.; Reifenrath, J.; Lenarz, T.; Durisin, M. Biocompatibility of MgF2-coated MgNd2 specimens in contact with mucosa of the nasal sinus—A long term study. *Acta Biomater*. 2015, 18, 249–261. [CrossRef] [PubMed]
- Durisin, M.; Reifenrath, J.; Weber, C.M.; Eifler, R.; Maier, H.J.; Lenarz, T.; Seitz, J.M. Biodegradable nasal stents (MgF2 -coated Mg-2 wt %Nd alloy)-A long-term in vivo study. *J. Biomed. Mater Res. B Appl. Biomater.* 2017, 105, 350–365. [CrossRef] [PubMed]
- Behring, J.; Junker, R.; Walboomers, X.F.; Chessnut, B.; Jansen, J.A. Toward guided tissue and bone regeneration: Morphology, attachment, proliferation, and migration of cells cultured on collagen barrier membranes. A systematic review. *Odontology* 2008, *96*, 1–11. [CrossRef]
- 57. Bunyaratavej, P.; Wang, H.L. Collagen membranes: A review. J. Periodontol. 2001, 72, 215–229. [CrossRef]
- 58. Guda, T.; Walker, J.A.; Singleton, B.M.; Hernandez, J.W.; Son, J.S.; Kim, S.G.; Oh, D.S.; Appleford, M.R.; Ong, J.L.; Wenke, J.C. Guided bone regeneration in long-bone defects with a structural hydroxyapatite graft and collagen membrane. *Tissue Eng. Part A* **2013**, *19*, 1879–1888. [CrossRef]
- 59. Hoornaert, A.; d'Arros, C.; Heymann, M.F.; Layrolle, P. Biocompatibility, resorption and biofunctionality of a new synthetic biodegradable membrane for guided bone regeneration. *Biomed. Mater.* **2016**, *11*, 045012. [CrossRef]

- Byun, S.H.; Lim, H.K.; Kim, S.M.; Lee, S.M.; Kim, H.E.; Lee, J.H. The Bioresorption and Guided Bone Regeneration of Absorbable Hydroxyapatite-Coated Magnesium Mesh. J. Craniofac. Surg. 2017, 28, 518–523. [CrossRef]
- 61. Fischerauer, S.F.; Kraus, T.; Wu, X.; Tangl, S.; Sorantin, E.; Hanzi, A.C.; Loffler, J.F.; Uggowitzer, P.J.; Weinberg, A.M. In vivo degradation performance of micro-arc-oxidized magnesium implants: A micro-CT study in rats. *Acta Biomater.* **2013**, *9*, 5411–5420. [CrossRef] [PubMed]
- 62. Agarwal, S.; Curtin, J.; Duffy, B.; Jaiswal, S. Biodegradable magnesium alloys for orthopaedic applications: A review on corrosion, biocompatibility and surface modifications. *Mater. Sci. Eng. C Mater. Biol. Appl.* **2016**, *68*, 948–963. [CrossRef] [PubMed]
- 63. Jung, O.; Smeets, R.; Kopp, A.; Porchetta, D.; Hiester, P.; Heiland, M.; Friedrich, R.E.; Precht, C.; Hanken, H.; Grobe, A.; et al. PEO-generated Surfaces Support Attachment and Growth of Cells In Vitro with No Additional Benefit for Micro-roughness in Sa (0.2–4 μm). *In Vivo* 2016, *30*, 27–33. [PubMed]
- 64. Sun, J.; Wang, J.; Jiang, H.; Chen, M.; Bi, Y.; Liu, D. In vivo comparative property study of the bioactivity of coated Mg-3Zn-0.8Zr alloy. *Mater Sci. Eng. C Mater Biol. Appl.* **2013**, *33*, 3263–3272. [CrossRef] [PubMed]
- 65. Kastellorizios, M.; Tipnis, N.; Burgess, D.J. Foreign Body Reaction to Subcutaneous Implants. *Adv. Exp. Med. Biol.* **2015**, *865*, 93–108. [CrossRef]
- Gruionu, G.; Stone, A.L.; Schwartz, M.A.; Hoying, J.B.; Williams, S.K. Encapsulation of ePTFE in prevascularized collagen leads to peri-implant vascularization with reduced inflammation. *J. Biomed. Mater Res. A* 2010, *95*, 811–818. [CrossRef]
- 67. Klopfleisch, R.; Jung, F. The pathology of the foreign body reaction against biomaterials. *J. Biomed. Mater. Res. A* 2017, *105*, 927–940. [CrossRef]
- 68. Major, M.R.; Wong, V.W.; Nelson, E.R.; Longaker, M.T.; Gurtner, G.C. The foreign body response: At the interface of surgery and bioengineering. *Plast. Reconstr. Surg.* **2015**, *135*, 1489–1498. [CrossRef]
- 69. DiEgidio, P.; Friedman, H.I.; Gourdie, R.G.; Riley, A.E.; Yost, M.J.; Goodwin, R.L. Biomedical implant capsule formation: Lessons learned and the road ahead. *Ann. Plast. Surg.* **2014**, *73*, 451–460. [CrossRef]
- 70. Ghanaati, S.; Unger, R.E.; Webber, M.J.; Barbeck, M.; Orth, C.; Kirkpatrick, J.A.; Booms, P.; Motta, A.; Migliaresi, C.; Sader, R.A.; et al. Scaffold vascularization in vivo driven by primary human osteoblasts in concert with host inflammatory cells. *Biomaterials* 2011, *32*, 8150–8160. [CrossRef]
- 71. Klopfleisch, R. Macrophage reaction against biomaterials in the mouse model—Phenotypes, functions and markers. *Acta Biomater.* **2016**, *43*, 3–13. [CrossRef] [PubMed]
- Abshagen, K.; Schrodi, I.; Gerber, T.; Vollmar, B. In vivo analysis of biocompatibility and vascularization of the synthetic bone grafting substitute NanoBone. *J. Biomed. Mater. Res. A* 2009, *91*, 557–566. [CrossRef] [PubMed]
- 73. Perumal, G.; Ramasamy, B.; Doble, M.J.C.; Biointerfaces, S.B. Nanostructure coated AZ31 magnesium cylindrical mesh cage for potential long bone segmental defect repair-in vitro degradation and cytocompatibility studies. *Colloids Surf. B Biointerfaces* **2018**, *172*, 690–698. [CrossRef] [PubMed]
- Li, Z.; Yang, W.; Yu, Q.; Wu, Y.; Wang, D.; Liang, J.; Zhou, F.J.L. New Method for the Corrosion Resistance of AZ31 Mg Alloy with a Porous Micro-Arc Oxidation Membrane as an Ionic Corrosion Inhibitor Container. *Langmuir* 2018, 35, 1134–1145. [CrossRef] [PubMed]
- Zomorodian, A.; Garcia, M.P.; Moura e Silva, T.; Fernandes, J.C.; Fernandes, M.H.; Montemor, M.F. Corrosion resistance of a composite polymeric coating applied on biodegradable AZ31 magnesium alloy. *Acta Biomater.* 2013, *9*, 8660–8670. [CrossRef] [PubMed]
- 76. Gomes, P.S.; Zomorodian, A.; Kwiatkowski, L.; Lutze, R.; Balkowiec, A.; Colaço, B.; Pinheiro, V.; Fernandes, J.C.; Montemor, M.F.; Fernandes, M.H.J.B.M. In vivo assessment of a new multifunctional coating architecture for improved Mg alloy biocompatibility. *Biomed. Mater.* 2016, 11, 045007. [CrossRef]
- 77. Neacsu, P.; Staras, A.I.; Voicu, S.I.; Ionascu, I.; Soare, T.; Uzun, S.; Cojocaru, V.D.; Pandele, A.M.; Croitoru, S.M.; Miculescu, F.; et al. Characterization and In Vitro and In Vivo Assessment of a Novel Cellulose Acetate-Coated Mg-Based Alloy for Orthopedic Applications. *Materials* **2017**, *10*, 686. [CrossRef]
- 78. Jiang, Y.; Wang, B.; Jia, Z.; Lu, X.; Fang, L.; Wang, K.; Ren, F. Polydopamine mediated assembly of hydroxyapatite nanoparticles and bone morphogenetic protein-2 on magnesium alloys for enhanced corrosion resistance and bone regeneration. *Biomed. Mater.* **2017**, *105*, 2750–2761. [CrossRef]

- 79. Li, G.; Zhang, L.; Wang, L.; Yuan, G.; Dai, K.; Pei, J.; Hao, Y. Dual modulation of bone formation and resorption with zoledronic acid-loaded biodegradable magnesium alloy implants improves osteoporotic fracture healing: An in vitro and in vivo study. *Acta Biomater.* **2018**, *65*, 486–500. [CrossRef]
- 80. Jiang, H.; Wang, J.; Chen, M.; Liu, D. Biological activity evaluation of magnesium fluoride coated Mg-Zn-Zr alloy in vivo. *Mater. Sci. Eng. C Mater. Biol. Appl.* **2017**, *75*, 1068–1074. [CrossRef]
- Li, Z.; Shizhao, S.; Chen, M.; Fahlman, B.D.; Liu, D.; Bi, H. In vitro and in vivo corrosion, mechanical properties and biocompatibility evaluation of MgF2-coated Mg-Zn-Zr alloy as cancellous screws. *Mater. Sci. Eng. C Mater. Biol. Appl.* 2017, 75, 1268–1280. [CrossRef] [PubMed]
- 82. Donath, K.; Breuner, G. A method for the study of undecalcified bones and teeth with attached soft tissues. The Sage-Schliff (sawing and grinding) technique. *J. Oral Pathol.* **1982**, *11*, 318–326. [CrossRef] [PubMed]
- Hahn, M.; Vogel, M.; Delling, G. Undecalcified preparation of bone tissue: Report of technical experience and development of new methods. *Virchows Arch A Pathol. Anat. Histopathol.* 1991, 418, 1–7. [CrossRef] [PubMed]



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#### Zusammenfassende Darstellung der Publikationspromotion

Die Verweise auf Abbildungen (engl.: *Figures*) beziehen sich auf die anhängige Originalpublikation.

## 1. Einleitung

Knöcherne Defekte in der regenerativen bzw. Mund-, Kiefer- und Gesichtschirurgie und Zahnmedizin, beispielsweise bedingt durch Missbildungen, Resektionen von tumorösem Gewebe, Degeneration oder Unfällen, können teilweise extensive Rekonstruktionen erforderlich machen. Bis dato stellt die autologe Knochentransplantation in der Therapie den Goldstandard dar (Akbay and Aydogan, 2014, Milinkovic and Cordaro, 2014, Frodel, 1999). Die sogenannte GBR/GTR (guided bone/tissue regeneration) bezeichnet hierbei eine Technik, bei der die Defekte im Kieferkammbereich nach Knochentransplantation durch resorbierbare oder nicht-resorbierbare Membranen abgedeckt werden, um das schneller wachsende Weichgewebe aus dem Defekt fernzuhalten und Knochenregeneration zu ermöglichen (Basler et al., 2018, Milinkovic and Cordaro, 2014, Smeets et al., 2014). Bei Verwendung einer klassischen resorbierbaren Kollagenmembran kann es allerdings zu einem Kollaps der Membran in den Defekt kommen, was die Knochenneubildung beeinträchtigt (Sheikh et al., 2017, Tolstunov et al., 2019). Die Entwicklung volumenstabiler nicht-resorbierbarer titanverstärkter Membranen konnte dieses Problem beheben, erfordert jedoch eine operative Entfernung nach erfolgter Regeneration (Elgali et al., 2017, Korzinskas et al., 2018b). Zur Deckung ausgedehnterer Defekte, beispielsweise nach Durchführung kontinuitätserhaltender Kieferteilresektionen, nach Resektion von Zysten oder bei ausgeprägten Atrophien des Kieferkamms, können kortikospongiöse Beckenkammtransplantate verwendet werden (Sindet-Pedersen and Enemark, 1990). Dieses Transplantationsverfahren stellt allerdings bei höherer Invasivität eine größere Belastung für den Patienten mit potenziell erhöhter Entnahmemorbidität dar (Dimitriou et al., 2011, Rawashdeh and Telfah, 2008). Solcherlei medizinische Maßnahmen bedeuten hohe volkswirtschaftliche Kosten und belasten das Gesundheitssystem zudem kapazitativ (Farkas and Becks, 2005). Zwecks Knochenersatz und Fixierung von Knochenfragmenten nach Frakturierung werden auch metallene Implantate, zum Beispiel aus Titan, eingesetzt. Auch in diesem Fall muss das implantierte Material später wieder entfernt werden, was nebst vermehrter Kosten ebenfalls mit vermeidbaren Operationsrisiken und mit patientenseitiger psychischer Belastung einhergeht (Jacobi-Gresser et al., 2013). Die Behandlung von Knochendefekten mit einem resorbierbaren Material könnte die Problematik des Zweiteingriffes für den Patienten beheben und eine Kostenersparnis erwirken. Darüber hinaus wären auch Knochenersatzmaterialen (KEM) zum Ersatz fehlender Kontinuität aus einem volumenstabilen und resorbierbaren Material, welches eigens Knochenbildung induziert, möglich und überaus vorteilhaft (Staiger et al., 2006).

Gute Biokompatibilität und eine knochenähnliche Stabilität machen Magnesium (Mg) in der Theorie zu einer idealen Basis für ein solches Implantatmaterial (Staiger et al., 2006). Dieses könnte bei potenziell gleichwertigen Ergebnissen, verglichen mit titanbasierten Strukturen, zur Fabrikation von Stützstrukturen für Barrieremembranen verwendet werden (Zheng et al., 2014, Staiger et al., 2006, Zhao et al., 2017). Die schnelle Degradation von reinem Magnesium hat allerdings bisher die klinische Verwendung als KEM oder Stützstruktur für Membranen aufgrund von vorzeitigem Funktionsverlust, starker Wasserstoffgasbildung mit Gefahr von Emphysembildung und Nekrosegefahr bei Alkalisierung verhindert (Staiger et al., 2006, Hornberger et al., 2012).

Ziel des Projekts war die Entwicklung einer neuartigen volumenstabilen Barrieremembran, welche die obig beschriebenen Probleme löst, Magnesium als Implantatmaterial für die Herstellung von Stützstrukturen für Barrieremembranen verwendbar macht und die vorteilhaften Charakteristika von resorbierbaren und nicht-resorbierbaren Membranen mit Stützstrukturen aus Titan in sich vereint. Dazu wurde eine porkine Kollagenmembran (Jason<sup>®</sup> Membran, botiss biomaterials GmbH, Zossen) mit einem Magnesiumgitter verstärkt. Um die Korrosion des Magnesiums zu verzögern, wurde in einer Testgruppe das Gitter zusätzlich mittels Immersionsbeschichtung mit Fluorwasserstoff (HF) behandelt, was eine reine, dünne Magnesiumfluorid (MgF<sub>2</sub>)-Schicht zum Ergebnis hatte.

Um valide Aussagen über die Biokompatibilität, Degradation und die Gewebereaktion der Membran *in vitro* und *in vivo* zu erhalten, wurden die Membranen mittels etablierter *in vitro* Methoden und standardisiert im Tiermodell untersucht. Hierzu wurden die Membranen auf 8 mm große Defekte in den Kalotten 6 Monate alter Kaninchen positioniert. Nach 6, 12 und 18 Wochen erfolgte die histologische Aufarbeitung und Untersuchung der Explantate mittels digitaler Volumentomographie (DVT), Kontaktradiographie und durch histologische Untersuchungen hinsichtlich Knochenregeneration, Degradationsverhalten der Membranen, sowie histopathologische Analysen der Reaktion des umliegenden Gewebes und der Wasserstoffgasentwicklung.

## 2. Material und Methoden

## 2.1 Vorbehandlung des Biomaterials

Magnesiumbögen der Legierung AZ31 (biotrics bioimplants GmbH, Berlin, Deutschland) wurden aufgereinigt und durch Immersion in HF-Lösung beschichtet. Der Zuschnitt für die Einbettung in der Kollagenmembran (Jason<sup>®</sup> Membran, botiss biomaterials GmbH, Zossen, Deutschland) erfolgte per Laser.

## 2.2 Zytokompatibilitätsanalyse

Die durchgeführte Zytokompatibilitätsanalyse ist bereits ausführlich beschrieben (Jung et al., 2016, Jung et al., 2019, Jung et al., 2015) und der Versuchsaufbau wird im Folgenden kurz erläutert.

## 2.2.1 Referenzmaterialien

RM-A (Hatano-Forschungsinstitut, Food and Drug Safety Center, Japan), als Positivkontrolle, Wako-Kunststofffolien (Wako Pure Chemical Industries, Ltd., Osaka, Japan, Kat. Nr. 160-08893), als ungiftiges Kontrollmaterial und Titan wurden analog zu den Materialproben hergestellt und sterilisiert.

## 2.2.2 Zellen und Zellkultur

L-929- und MC3T3-Zellen wurden von der European Collection of Cell Culture, ECACC (Salisbury, UK) erworben. Die Zellen wurden in Zellkulturmedium (MEM (Minimum Essential Medium), ergänzt mit 10% fetalem Rinderserum, Penicillin / Streptomycin (alle von Life Technologies, Carlsbad, CA, USA) und L-Glutamin (Sigma-Aldrich, St. Louis, MO, USA) bis zu einer Endkonzentration von 4 mM) unter Zellkulturbedingungen kultiviert.

## 2.2.3 Extraktanalyse

## 2.2.3.1 Extraktion

Test- und Kontrollproben wurden 72 h lang in Zellkulturmedium unter Zellkulturbedingungen extrahiert. Zellkulturmedium allein wurde unter identischen Bedingungen als negativer Kontrollextrakt inkubiert. Nach der Entnahme der Proben wurden die verbleibenden Extrakte zentrifugiert und die Überstände wie nachstehend beschrieben verwendet.

## 2.2.3.2 Assay-Verfahren

96-Well Platten wurden mit 1 ×  $10^4$  L929-Zellen / Well in 100 µl Zellkulturmedium beimpft und 24 h unter Zellkulturbedingungen inkubiert. Danach wurde das Zellkulturmedium verworfen und 100 µl Extrakt in jedes Well gegeben. Die Zellen wurden für 24 Stunden weiter inkubiert und dann den BrdU- und XTT-Tests unterzogen, während die Überstände dem Lactatdehydrogenase (LDH) -Test zugeführt wurden. Identische Assays, jedoch ohne Zellen, wurden für alle Extrakte als Kontrolle für Assayinterferenzen durchgeführt.

## 2.2.3.3 Bromdesoxyuridin (BrdU)-Assay

Das BrdU (colorimetric) Testkit (Roche Diagnostics, Mannheim, Deutschland) wurde gemäß den Anweisungen des Herstellers verwendet. Die Zellen wurden 2 h unter Zellkulturbedingungen mit BrdU markiert und anschließend 30 min bei Raumtemperatur mit FixDenat-Reagenz fixiert. Dann wurden die fixierten Zellen 1 Stunde mit Anti-BrdU-Peroxidase (POD)-Antikörper inkubiert und mit Waschpuffer gewaschen. Die Immunkomplexe wurden nach einer anschließenden Substratreaktion mit Tetramethylbenzidin (TMB) und darauffolgender Zugabe von 25  $\mu$ l 1 M H<sub>2</sub>SO<sub>4</sub>, zum Stoppen der Reaktion, mit einem Raster-Multiwell-Spektrophotometer nachgewiesen.

## 2.2.3.4 Natrium 3,3'- [1(Phenylamino)carbonyl]-3,4-Tetrazolium]-3is (4-Methoxy-6-Nitro) Benzolsulfonsäurehydrat (XTT)-Assay

Das Cell Proliferation Kit II (Roche Diagnostics, Mannheim, Deutschland) wurde gemäß den Anweisungen des Herstellers verwendet. Das Elektronenkopplungsreagenz wurde mit XTT-Markierungsreagenz (1:50 Verdünnung) gemischt und 50 µl der Mischung wurden zu den Zellen gegeben. Nach 4-stündiger Inkubation unter Zellkulturbedingungen wurde die Substratumwandlung quantifiziert, indem die Absorption von 100 µl-Proben in einer neuen 96-Well-Platte unter Verwendung eines Raster-Multiwell-Spektrophotometers ermittelt wurde.

## 2.2.3.5 Laktatdehydrogenase (LDH) -Assay:

Das LDH-Cytotoxicity Assay Kit II (BioVision, Milpitas, CA, USA) wurde gemäß den Anweisungen des Herstellers verwendet. 10 µl der Zellüberstände wurden mit 100 µl LDH-Reaktionsgemisch 30 min bei Raumtemperatur inkubiert. Nach Zugabe der Stopplösung wurden die Extinktionen mit einem Raster-Multiwell-Spektrophotometer gemessen.

## 2.2.4 Lebend-Tot-Färbung:

Mg-Proben und Kontrollen wurden mit L929- und MC3T3-Zellen in 12 Well-Platten besät. Die Untersuchungen wurden nach 24-stündiger Inkubation unter Zellkulturbedingungen durchgeführt. Es wurden 60 µl/ml Medium Propidiumiodid (PI)-Stammlösung und 500 µl/ml Medium frische Fluoresceindiacetat (FDA)-Lösung verwendet. Nach einer kurzen Inkubation von 3 min bei Raumtemperatur wurden die Proben mit vorgewärmtem PBS gespült und sofort mit einem Fluoreszenzmikroskop (Nikon ECLIPSE Ti-S / L100, Nikon GmbH, Düsseldorf, Deutschland) untersucht. Die Bilder wurden in 4-, 10- und 20-facher Vergrößerung

aufgenommen. Die Zellen wurden mit der Software ImageJ gezählt. Für jedes Material wurden Mittelwerte und Standardabweichungen ermittelt.

## 2.3 Tiermodell und Anästhesie

Bei den verwendeten Versuchstieren handelt es sich um ca. 12 Wochen alte New Zealand White Rabbits mit einem Gewicht von 2,2-2,9 kg (Charles River Laboratories, Inc., Wilmington, USA). Zur Narkoseeinleitung wurde den Versuchstieren 5 mg/kg Körpergewicht Xylazin (Rompun<sup>®</sup> 2%, 20 mg/ml, Anästhetikum, Analgetikum, Muskelrelaxanz, Bayer Vital GmbH, Leverkusen, Deutschland) und 70 mg/kg Körpergewicht Ketamin (Ketamin<sup>®</sup> 10%, 100 mg/ml, Anästhetikum, Analgetikum, Bela-pharm GmbH & Co. KG, Vechta, Deutschland) subkutan verabreicht. Bei nicht ausreichender initialer Dosis wurde in 0,1 ml-Schritten aus einem 0,3 ml Ketamin-/0,7 ml Xylazin-Gemisch (1,0 ml) nachdosiert. Nach erfolgter Narkoseeinleitung wurden die Tiere mittels Larynxtubus (v-gel<sup>®</sup>, Millpledge Veterinary, UK) intubiert und beatmet. Alle Tiere erhielten zusätzlich intravenös ein Opioid. Bei 11 Tieren kamen 0,1 ml Buprenorphin (Buprenovet<sup>®</sup>, 0,3 mg/ml, Bayer Vital GmbH, Leverkusen, Deutschland), bei 7 Tieren Fentanyl (Fentanyl<sup>®</sup>-Janssen, 0,5 g/10 ml, Janssen Cilag GmbH, Neuss, Deutschland) über einen Perfusor mit einer Perfusionsgeschwindigkeit von 2,5-25 µg/kg/h zum Einsatz. Als Volumentherapie erhielten die Versuchstiere ca. 100 ml einer Stereofundin<sup>®</sup>-Infusionslösung (B. Braun Melsungen AG, Melsungen, Deutschland).

## 2.4 Implantation der zu untersuchenden Membranen

Nach lokaler Rasur der Schädeldecke, Hautdesinfektion mittels Cutasept<sup>®</sup> F (BODE Chemie GmbH, Hamburg, Deutschland) und subkutaner Injektion von mit Adrenalin versetztem Lokalanästhetikum wurde die Kopfhaut unter sterilen Bedingungen auf ca. 5 cm entlang der Mittellinie inzidiert. Nun wurden bilateral parietal, sowie bilateral frontal der Mittellinie unter steter Spülung mit Natriumchlorid-Lösung (NaCl 0,9%, B. Braun Melsungen AG, Melsungen, Deutschland) vier standardisierte runde Trepanationen durch die komplette Kalotte mit einem Durchmesser von 8 mm mittels Trepanbohrer (Ref. Nr. 330205486001070, Meisinger, Neuss, Deutschland) angelegt.

Gemäß eines randomisierten Versuchsprotokolls wurde pro Tier jeweils ein Defekt mit einer porkinen Kollagenmembran (Jason<sup>®</sup> Membran, botiss biomaterials GmbH, Zossen, Deutschland), einer HF-beschichteten Magnesium-verstärkten Kollagenmembran und einer unbeschichteten Magnesium-verstärkten Membran abgedeckt. Ein weiterer Defekt blieb als Kontrolle unbedeckt. Abschließend erfolgte die Rückverlagerung und der Verschluss des Periosts durch Einzelknopfnähte (4-0 Vicryl<sup>®</sup>, Ethicon Inc., USA), die Haut wurde mit Klammern geschlossen.

## 2.5 Euthanasie

Die Narkoseeinleitung erfolgte mit obig genannter Medikation. Die Euthanasierung wurde mit intrakardialer Injektion von 3-5 ml T61 (T61, MSD Tiergesundheit, Unterschleißheim, Deutschland) durchgeführt. Nach Feststellung des Todes wurde der Kopf mit den zu untersuchenden Membranen chirurgisch entfernt und in einer 4%-Formalinlösung fixiert.

## 2.6 Histologische Aufarbeitung

Zwecks histopathologischer und histomorphometrischer Analyse wurden die Präparate als Paraffin-Schnitte jeweils mit Färbungen nach Masson-Goldner, von Kossa und Toluidinblau gefärbt. Darüber hinaus wurden Schliffpräparate der in Kunststoff (Technovit 7200 VLC, Heraeus-Kulzer GmbH & Co. KG, Wehrheim, Deutschland) eingebetteten Zuschnitte mit einer Dicke von 30 µm angefertigt.

## 2.7 Bestimmung der Knochenregeneration mittels DVT und Kontaktradiographie

Die Präparatköpfe wurden zur Beurteilung mittels digitaler Volumentomographie (DVT) (90 kV, Promax, Planmeca, Helsinki, Finnland) und Kontaktradiographie (26 kV, LX-60, Faxitron X-Ray LLC, Tucson, USA) hochauflösend abgebildet.

Hierzu wurde der DVT-Datensatz des jeweiligen Tieres im DICOM-Format in das Programm "Horos" (https://www.horosproject.org) geladen und die Dicke der abgebildeten Schichten auf 6,45 mm erhöht. Nach Aufsuchen der Kalottenebene wurde ein 8 mm messender Eichmaßstab eingefügt und das Bild im TIF-Format exportiert. Dieses Bild wurde in das Programm "ImageJ" (National Institute of Health, https://imagej.nih.gov/ij/) geladen. Die Skalierung für Pixel pro Strecke erfolgte anhand des eingefügten Eichmaßstabs. Mit einer kreisförmigen 8 mm durchmessenden Auswahl wurde entlang der originalen Bohrmarken ein Ausschnitt des jeweiligen Loches erstellt.

Für die Kontaktradiographie wurde der Knochenzuwachs ebenfalls mit Hilfe des Programms "ImageJ" bestimmt. Hierzu wurde das Bild des jeweiligen Tieres im TIF-Format in das Programm geladen und es erfolgte die Skalierung für Pixel pro Strecke mittels Eichmaßstab auf dem Originalbild. Danach wurde eine kreisförmige Auswahl mit 8 mm Durchmesser, entsprechend der Originaltrepanation, ausgewählt und ein Ausschnitt des jeweiligen Loches entlang der Bohrmarken erstellt (Fig. 9).

Durch Festlegung eines Schwellenwertes bei einer Farbtiefe von 8-Bit wurden knöcherne Strukturen von nicht bewachsenen Arealen abgegrenzt und letztlich mithilfe eines Auswahltools die Gesamtgröße des Loches und die Anzahl der knöchern regenerierten Areale in Pixeln gemessen. Dividiert man die Anzahl der Pixel für die regenerierte Fläche durch die Gesamtgröße des Loches, ergibt sich die dezimale Knochenregeneration pro Loch (Fig. 9). 2.8 Histomorphometrische und -pathologische Untersuchungen der histologischen Präparate Die in Toluidinblau gefärbten Dünnschliffpräparate wurden bei 20-facher Vergrößerung mit Hilfe eines Messokulars und eines geeigneten Eichmaßstabs hinsichtlich des Anteils neugebildeter Knochenstrukturen in Relation zur Defektgröße unter dem Mikroskop (Zeiss 47 52 69, Carl Zeiss Microscopy GmbH, Jena, Deutschland) gemessen. Darüber hinaus erfolgte eine Messung der mit den magnesiumbeinhaltenden Membranen assoziierten Wasserstoffgasbildung durch digitalisierte Scans, die unter Zuhilfenahme eines Mikroskops (Axio Scope.A1, Carl Zeiss Microscopy GmbH, Jena, Deutschland) und der dazugehörigen Software (ZEN core, Zeiss, Tokyo, Japan) bestimmt wurde.

Mit dem letztgenannten Mikroskop erfolgte ebenfalls die histopathologische Analyse hinsichtlich der Material-assoziierten Gewebereaktion.

## 2.9 Statistische Auswertung

Mittelwerte und Standardabweichungen der mit dem Mikroskop und dem Auswertungsprogramm "ImageJ" ermittelten Daten werden im Ergebnisteil dargestellt. Es erfolgte eine ANOVA mittels Bonferroni Post-hoc-Test (SPSS 24, IBM, Armonk, New York, USA). Das Signifikanzniveau wurde bei  $p \le 0,05$  definiert.

## 3. Ergebnisse

#### 3.1 Zytokompatibilitätsanalysen

Sowohl im XTT- als auch im BrdU-Test zeigte HF-behandeltes Magnesium eine gute Zytokompatibilität, während unbehandeltes Magnesium im toxischen Bereich lag (Fig. 4 A-C). Dabei unterschied sich HF-behandeltes Magnesium nicht signifikant von der Negativkontrolle im BrdU-Assay (p > 0,05) aber signifikant von der Negativkontrolle im XTT-Assay ( $p \le 0,05$ ). Sowohl die unbehandelten Magnesium- als auch die positiven Kontrollgruppen unterschieden sich signifikant von der negativen Kontrolle ( $p \le 0,05$ ). Im LDH-Assay befanden sich beide Magnesiumgruppen im toxischen Bereich, während sich nur die HF-behandelte Magnesiumgruppe im Vergleich zur Negativkontrolle nicht signifikant unterschied (p > 0,05). Die morphologische Beurteilung der Magnesium-Testproben nach 72-stündigem Extraktionstest ergab eine verstärkte Korrosion der unbehandelten Testproben, während die HF-behandelten Proben keine derartigen Anzeichen aufwiesen (Fig. 4 D).

In der Lebend-Tot-Färbung zeigte die mit HF behandelte Gruppe ähnliche Ergebnisse wie die Negativkontrolle, obwohl die Adhäsion der MC3T3-Zellen besser zu sein scheint (Fig. 5). Unbehandeltes Magnesium zeigte eine schlechte Zelladhäsion mit einigen roten toten Zellen.

## 3.2 Histopathologische Untersuchung

Fibroblastenakkumulation (schwarzer Pfeil, Fig. 6, 7) um Reste von unbehandeltem Magnesium (Mg, Fig. 6, 7) und HF-beschichtetem Magnesium (Mg<sup>#</sup>, Fig. 6, 7) waren zu allen Zeitpunkten und Wasserstoffgasansammlungen, sowie Bildung von fibrotischen Kapseln ( $\star$ , Fig. 6, 7) bis zu 12 Wochen sichtbar. Die Dicke der Kapseln unterlag lokalen Schwankungen, nahm jedoch wie die Gasbildung im Zeitverlauf ab. Wie im Folgenden beschrieben, war lediglich eine minimale abnehmende Entzündungsreaktion im umliegenden Gewebe bei beiden Gruppen sichtbar.

Nach 6 Wochen löste unbehandeltes Magnesium eine fibroseartige Gewebereaktion aus und wurde hauptsächlich durch Auflösung und kaum durch phagozytische Prozesse abgebaut. HFbeschichtetes Magnesium hingegen wurde hauptsächlich durch aktive Phagozytose von Monozyten resorbiert (gelber Pfeil, Fig. 6, 7), war aber ebenfalls von Fibrosebildung begleitet. Nach 12 Wochen war eine Verschiebung der Zersetzung von HF-beschichtetem Magnesium in Richtung einer nicht zellgetriebenen Auflösung, wie bei unbehandeltem Magnesium, sichtbar. Es zeigten sich deutlich weniger monozytische Zellen. Nach 18 Wochen war lediglich eine dünne Kapsel um kaum nachweisbare Gashohlräume und wenige Fibroblasten zu erkennen, was zu dem Schluss führte, dass nach anfänglichem Abbau der HF-Beschichtung bis zu einigen µm oder sogar direkter Kontakt zwischen Knochengewebe und Implantat nach 6 und 12 Wochen sind als hervorragende Biokompatibilität zu werten.

## 3.3 DVT

Es konnten keine statistisch signifikanten Unterschiede zwischen den Mengen an neu gebildetem Knochen in den untersuchten Gruppen gefunden werden (p > 0.05).

Nach 6 Wochen wurden 76,17 ± 9,50% neugebildeter Knochen in der Gruppe der HFbeschichteten Magnesiummembranen gefunden, was mit der Regeneration für unbeschichtete Mg-Membranen (75,69 ± 9,23%), der Kollagengruppe (77,86 ± 7,59%) und Kontrollgruppe (74,25 ± 7,13%) vergleichbar war. Nach 12 Wochen zeigten sich ähnliche Ergebnisse in der Gruppe der HF-beschichteten Magnesiummembranen (72,17 ± 11,05%), der unbeschichteten Membran (70,23 ± 7,49%), der Kollagengruppe (77,11 ± 8,50%) und der Kontrollgruppe (70,98 ± 11,50%). Nach 18 Wochen waren die Ergebnisse für die HFbeschichtete Magnesiummembran (75, 50)19,24%), die unbeschichtete ± Magnesiummembran (77,93 ± 9,94%), die Kollagengruppe (78,59 ± 13,88%) und die Kontrollgruppe (78,58 ± 12,19%) ebenfalls vergleichbar (Fig. 8B).

## 3.4 Kontaktradiographie

Es konnten keine statistisch signifikanten Unterschiede zwischen den Mengen an neu gebildetem Knochen in den untersuchten Gruppen gefunden werden (p > 0.05).

Nach 6 Wochen wurden vergleichbare Ergebnisse in der Gruppe der HF-beschichteten Magnesiummembranen (62,37 ± 13,77%), der unbeschichteten Magnesiummembranen (66,78 ± 5,11%), der Kollagengruppe (67,18 ± 14,29%) und der Kontrollgruppe gefunden (55,64 ± 12,98%). 12 Wochen nach der Implantation zeigten sich ähnliche Ergebnisse für die 16,42%), HF-beschichtete Magnesiummembran (64.26 ± die unbeschichtete Magnesiummembran (69,45 ± 10,80%), die Kollagengruppe (68,59 ± 12,90%) und die Kontrollgruppe (59,50 ± 19,61%). Nach 18 Wochen fanden sich ebenfalls vergleichbare Ergebnisse für die HF-beschichteten Membranen (74,07 ± 15,28%), die unbeschichteten Magnesiummembranen (69.85  $\pm$  5.61%), die Kollagengruppe (69.15  $\pm$  10.84%) und die Kontrollgruppe (70,49 ± 11,96%) (Fig. 8C).

## 3.5 Histomorphometrische Untersuchung

Es konnten keine statistisch signifikanten Unterschiede zwischen den Mengen an neu gebildetem Knochen in den untersuchten Gruppen gefunden werden (p > 0.05).

6 Wochen nach Implantation wurden 46.48  $\pm$  11.42% neugebildeter Knochen in der Gruppe der HF-beschichteten Magnesiummembranen gefunden, was mit der Regeneration für unbeschichtete Mg-Membranen (50.83  $\pm$  22.09%), der Kollagengruppe (49.40  $\pm$  17.55%) und

Kontrollgruppe (47.88  $\pm$  22.34%) vergleichbar war. Nach 12 Wochen waren ähnliche Ergebnisse in der Gruppe der HF-beschichteten Magnesiummembranen (41.21  $\pm$  29.17%), der unbeschichteten Membran (52.02  $\pm$  19.55%), der Kollagengruppe (46.59  $\pm$  18.34%) und der Kontrollgruppe (62.20  $\pm$  20.86%) erkennbar. 18 Wochen nach Implantation zeigten sich ebenfalls vergleichbare Ergebnisse für die HF-beschichtete Magnesiummembran (56.59  $\pm$  33.60%), die unbeschichtete Magnesiummembran (59.79  $\pm$  18.14%), die Kollagengruppe (74.39  $\pm$  14.40%) und die Kontrollgruppe (58.15  $\pm$  10.46%) (Fig. 8D).

## 3.6 Untersuchung der Wasserstoffgasbildung

Die quantitativen Messungen der Wasserstoffgasbildung ergaben 6 Wochen nach Implantation signifikant höhere Werte für unbeschichtetes (535 453  $\mu$ m<sup>2</sup> ± 323 998  $\mu$ m<sup>2</sup>) im Vergleich zu beschichtetem Magnesium (173 713  $\mu$ m<sup>2</sup> ± 129 697  $\mu$ m<sup>2</sup>) (p < 0,01) (Fig. 8A). 12 Wochen nach der Implantation zeigte die Gasbildung geringere Ergebnisse für die beschichteten (44 522  $\mu$ m<sup>2</sup> ± 44 457  $\mu$ m<sup>2</sup>) im Vergleich zu den unbeschichteten (131 299  $\mu$ m<sup>2</sup> ± 129 697  $\mu$ m<sup>2</sup>) magnesiumverstärkten Membranen. Die Verringerung der Gasblasengröße für unbeschichtete Membranen nach 12 Wochen war im Vergleich zu 6 Wochen signifikant geringer (p < 0,05) (Fig. 8A). Nach 18 Wochen konnte keine Gasbildung festgestellt werden. Insgesamt ist die Knochenregeneration durch die Gasbildung nicht beeinträchtigt (Fig. 8A).

## 4. Diskussion

Bioresorbierbare GBR/GTR-Membranen stellen im zahnmedizinischen Bereich eine gute Behandlungsoption dar, um Knochenregeneration zu ermöglichen und schneller wachsendes Weichgewebe aus dem Defekt fernzuhalten. Großvolumige Defekte können jedoch volumenstabile Membranen erfordern (Tolstunov et al., 2019, Sheikh et al., 2017). Bislang bestehen die verfügbaren volumenstabilen Membranen aus nicht resorbierbaren Materialien und müssen nach der Therapie chirurgisch entfernt werden. In diesem Zusammenhang wurde mehrfach gezeigt, dass Magnesium aufgrund seiner biokompatiblen, biologisch abbaubaren und mechanischen Eigenschaften ein günstiges Biomaterial für die Entwicklung stabilisierender Strukturen darstellt (Jung et al., 2015, Jung et al., 2019, Witte et al., 2005, Witte et al., 2006, Zhao et al., 2017, Ding, 2016). Als vielversprechende Oberflächenmodifikation für die Verzögerung des vorzeitigen Abbaus und damit der Freisetzung von Wasserstoff wurden HF-Beschichtungen und die daraus resultierenden MgF2-Schichten bereits für ein breiteres Anwendungsspektrum untersucht, das von orthopädischen (Yu et al., 2017) bis zu kardiovaskulären Anwendungen reicht (Mao et al., 2013, Zhao et al., 2014). Hierbei konnte eine allgemein höhere Korrosionsbeständigkeit festgestellt werden (Makkar et al., 2018, Liu et al., 2018, Kang et al., 2016, Zhao et al., 2014, Mao et al., 2013, Zhang et al., 2014, Yu et al., 2017).

Ziel dieser Studie war die *in vitro* und *in vivo* Untersuchung einer volumenstabilen GBR/GTR-Membran. Hierzu wurde eine native Kollagenmembran mit einem HF-behandelten Magnesiumgitter zur Kontrolle der Wasserstoffentwicklung und des Abbauprozesses kombiniert.

*In vitro* zeigte die HF-behandelte Membran eine vielversprechende Zytokompatibilität im Extrakt- und Direkttest (Lebend-Tot-Färbung), während die unbehandelte Membran in allen Extrakttests im toxischen Bereich lag. Im LDH-Assay war auch die HF-Membran toxisch. Der Unterschied zur Negativkontrolle war jedoch nicht signifikant, was für eine geringe Toxizität spricht. In den direkten Tests zeigten beide Magnesiumvarianten hauptsächlich lebende Zellen mit besserer Adhäsion für das HF-behandelte Magnesium. Unsere *in-vitro*-Ergebnisse zeigten, wie andere Studien, dass die HF-Behandlung von Magnesium zu einer verbesserten Zytokompatibilität, Adhäsion und Korrosionsbeständigkeit führt (Kang et al., 2016, Mao et al., 2013).

*In vivo* war insgesamt die Knochenregeneration für alle Testgruppen im Zeitverlauf ähnlich (*p* > 0,05). Die HF-Beschichtung wurde von mononukleären Phagozyten bis zu 12 Wochen abgebaut, während die unbeschichteten Magnesiumnetze lediglich eine fibroseartige Gewebereaktion zeigten. Nach 18 Wochen war die Gewebereaktion in beiden Gruppen ähnlich. Der Abbau beider Mg-Membranen war mit abnehmender Gasbildung bis zu 12 Wochen verbunden, die die Knochenregeneration nicht störte. Nach 18 Wochen konnte kein

Wasserstoffgas mehr festgestellt werden. Die HF-beschichteten Magnesiumgitter verursachten interessanterweise bis zu 6 Wochen nach der Implantation signifikant geringere Gasbildung.

Unsere Ergebnisse zeigen, dass die magnesiumverstärkte Membran mit der nativen Kollagenmembran oder Leerkontrolle vergleichbare regenerative Eigenschaften besitzt. In einem gleichartigen Versuchsaufbau haben Byun et al. ein mit Hydroxyapatit (HA) beschichtetes Magnesiumgitter in einem Rattenmodell untersucht (Byun et al., 2017). Neben dem zellulären Abbau von Magnesium konnte keine Wasserstoffgasfreisetzung nachgewiesen werden, was höchstwahrscheinlich auf die unterschiedliche Beschichtung oder die schnelle Einkapselung und Integration des HA in das umgebende Gewebe zurückzuführen ist (Byun et al., 2017). Während die HF-Beschichtung die Gashohlräume bis zu 12 Wochen reduziert, könnten HA-Beschichtungen oder andere Techniken in der Lage sein, Wasserstoffgas zu verhindern, zu verzögern oder zu absorbieren (Byun et al., 2017). In Bezug auf die Knochenneubildung wurden keine Unterschiede zwischen allen Gruppen festgestellt, weder in dieser Studie noch in der Arbeit von Byun et al. (Byun et al., 2017). Hinsichtlich des mikroskopischen Erscheinungsbildes des neu gebildeten Gewebes führten HA-beschichtete Gitter zu gut organisierten Defektbereichen, während die leeren Kontrollen verformt und mit fehlerhaftem kollagenem Gewebe gefüllt zu sein schienen. In unserer Studie waren jedoch keine derartigen qualitativen Unterschiede erkennbar, die möglicherweise auf die Verwendung eines anderen Tiermodells oder die Verwendung von Kollagen zur Einbettung zurückzuführen sind (Byun et al., 2017). In beiden Studien löste die Implantation der Magnesiummembranen eine entzündliche Gewebereaktion aus, die von Byun et al. nach 18 Wochen als vermehrt beschrieben wurde (Byun et al., 2017). In unserer Studie zeigten sowohl beschichtete als auch unbeschichtete Membranen nur eine geringfügige Entzündung, die mit der Zeit abnahm und durch mononukleäre Phagozytose der HF-beschichteten Membran und der Bildung von fibrotischen Kapseln sowie durch Zersetzung der unbehandelten Mg-Membran definiert wurde. Leider wird die Zellzusammensetzung, die an der Zersetzung des Implantats beteiligt ist, von Byun et al. nicht weiter erwähnt (Byun et al., 2017). In unserer Studie kann die Angleichung der Abbaumuster nach 12 Wochen durch den fortschreitenden Abbau der HF-Schicht erklärt werden.

Dies ist die erste Studie, die Ergebnisse von auf Magnesium basierenden Stützstrukturen in Kombination mit Kollagenmembranen für eine volumenstabile GBR/GTR-Therapie liefert. Trotzdem unterliegt diese Studie und ihre Ergebnisse diversen Einschränkungen und wirft weitere Fragen auf. Die Knochenregeneration war für alle Versuchsgruppen und Kontrollgruppen gleich, was die allgemeine Notwendigkeit von GBR/GTR-Membranen in Frage stellen kann. Die Werte für die Knochenregeneration waren jedoch für die angewendeten Methoden gemäß der Reihenfolge von DVT> Kontaktradiographie> und

Histologie unterschiedlich. Dies kann durch die Tatsache erklärt werden, dass Bildgebungsverfahren wie DVT und Radiographie größere Summenbilder aufnehmen, während für die histologische Analyse Schnitte untersucht wurden. Darüber hinaus stellt die ähnliche Dichte von Knochen und Magnesium eine Hürde bei der zweifelsfreien Zuordnung zu einer der beiden Materialien während der Analyse dar.

Daher sollte die Applikation der neuen Membran mit den Kontrollgruppen als nächstes in einem Defektmodell in Kombination mit Knochenersatzmaterialien erfolgen. Des Weiteren sollte ein Tiermodell genutzt werden, das dem menschlichen Kieferknochen ähnlich ist und eine Verlängerung des Untersuchungszeitraumes diskutiert werden, da kein gänzlicher Abbau des Magnesiums nach 18 Wochen stattfand. Ferner sollten die mechanischen Eigenschaften vor und möglichst nach der Implantation geprüft werden. Eine genauere Untersuchung entzündlicher Reaktionen (z. B. M1/M2) (Korzinskas et al., 2018a) und der Vergleich anderer Oberflächenmodifikationen wie Duplex- (MgF<sub>2</sub>/PCL) (Makkar et al., 2018), Nanokomposit-(Perumal et al., 2018) und Komposit-Beschichtungen (Liu et al., 2018, Li et al., 2018b) sowie weitere Magnesiumlegierungen (Makkar et al., 2018, Liu et al., 2018, Zhao et al., 2014, Mao et al., 2013, Zomorodian et al., 2017, Jiang et al., 2017a, Li et al., 2017) sollten erfolgen und versprechen interessante Ergebnisse für die Zukunft.

#### 5. Zusammenfassung

In der regenerativen bzw. Mund-, Kiefer- und Gesichtschirurgie und Zahnmedizin werden häufig Barrieremembranen eingesetzt, um ein vorzeitiges Einwachsen von Weichgewebe in knöcherne Defekte zu verhindern und Knochenregeneration ungehindert stattfinden zu lassen. Bei Verwendung einer klassischen resorbierbaren Kollagenmembran kann es allerdings zu einem Membrankollaps in den Defekt kommen, wodurch die Knochenneubildung beeinträchtigt wird. Die Entwicklung volumenstabiler nicht-resorbierbarer titanverstärkter Membranen konnte dieses Problem beheben, erfordert jedoch eine operative Entfernung nach erfolgter Regeneration. Diese Studie hatte zum Ziel, eine innovative volumenstabile bioresorbierbare HF-behandelte Magnesiumstützstruktur zu untersuchen, die in eine native resorbierbare Kollagenmembran eingebettet ist und damit die Vorteile von resorbierbaren und nicht-resorbierbaren Membranen in sich vereint.

Hierzu wurden in 18 New Zealand White Rabbits jeweils vier 8 mm durchmessende Kalottendefekte erzeuat und ieweils mit HF-behandelten Magnesiumeiner /Kollagenmembran, einer unbehandelten Magnesium-/Kollagenmembran, einer Kollagenmembran allein und einer Leerkontrolle bedeckt und nach 6, 12 und 18 Wochen per DVT, kontaktradiographisch sowie histologisch ausgewertet. Darüber hinaus wurden Zytokompatibilitätsassays durchgeführt.

Die HF-Behandlung konnte die Gasbildung durch Magnesiumdegradation signifikant verringern und nach 18 Wochen waren keine Gaseinschlüsse mehr nachweisbar. Histopathologisch wurden HF-behandelte Membranen durch mononukleäre Phagozytose bis zu 12 Wochen abgebaut, während die unbehandelten Mg-verstärkten Membranen eine fibroseartige Gewebereaktion auslösten. *In vitro* zeigte das HF-behandelte Magnesium eine erhöhte Zytokompatibilität. Es konnten keine signifikanten Unterschiede hinsichtlich der Knochenregeneration in allen Gruppen und für alle Messmethoden gezeigt werden, wobei die mittlere gemessene Knochenregeneration gemäß der Reihenfolge DVT > Kontaktradiographie > Histologie abnahm.

HF-behandelte Magnesiummembranen in Kombination mit einer Kollagenmembran stellen daher eine vergleichbare volumenstabile Alternative zu resorbierbaren und nichtresorbierbaren Membranvarianten dar. Weitere Oberflächenbehandlungen und Kombination Magnesiumlegierungen, sowie der Einsatz der Membran in mit Knochenersatzmaterialien sollten untersucht werden.

## 6. Summary

In dentistry, barrier membranes are often used to prevent premature ingrowth of soft tissue into bony defects and to allow for unhindered bone regeneration to proceed. A collapse of the membrane into the covered defect and subsequent impairment of bone formation is not uncommon with commercial collagen membranes. Titanium-reinforced membranes have been developed to counteract, however, surgical removal after regeneration is necessary. The aim of this study was to investigate an innovative volume-stable bioresorbable HF-treated magnesium mesh, embedded in a native resorbable collagen membrane that combines the benefits of resorbable and non-resorbable membranes.

In 18 New Zealand White Rabbits four calvarial defects of 8mm were generated, each covered with an HF-treated magnesium / collagen membrane, an untreated magnesium / collagen membrane, a collagen membrane alone and an empty control and were evaluated after 6, 12 and 18 weeks through DVT, contact radiography and histology. Furthermore, cytocompatibility tests were carried out.

HF-treatment significantly reduced gas formation due to magnesium degradation and after 18 weeks no gas cavities were further detectable. Histopathologically, HF-treated membranes were degraded by mononuclear phagocytosis for up to 12 weeks, while untreated magnesium reinforced membranes triggered a fibrosis-like tissue reaction. *In vitro*, HF-treated magnesium showed higher cytocompatibility. No significant differences in bone regeneration in all groups and for all methods were observed, with mean measured bone regeneration decreasing in the order of DVT> contact radiography> histology.

Thusly, HF-treated magnesium membranes combined with a collagen membrane represent an equal volume-stable alternative to resorbable and not-resorbable barrier membranes. Further surface treatments as well as magnesium alloys and the use of the membrane in combination with bone substitute materials should be investigated.

## Literaturverzeichnis

- AKBAY, E. & AYDOGAN, F. 2014. Reconstruction of isolated mandibular bone defects with non-vascularized corticocancellous bone autograft and graft viability. *Auris Nasus Larynx*, 41, 56-62.
- BASLER, T., NAENNI, N., SCHNEIDER, D., HÄMMERLE, C. H., JUNG, R. E. & THOMA, D. S. 2018. Randomized controlled clinical study assessing two membranes for guided bone regeneration of peri-implant bone defects: 3-year results. *Clinical oral implants research*.
- BYUN, S. H., LIM, H. K., KIM, S. M., LEE, S. M., KIM, H. E. & LEE, J. H. 2017. The Bioresorption and Guided Bone Regeneration of Absorbable Hydroxyapatite-Coated Magnesium Mesh. *J Craniofac Surg*, 28, 518-523.
- DIMITRIOU, R., MATALIOTAKIS, G. I., ANGOULES, A. G., KANAKARIS, N. K. & GIANNOUDIS, P. V. 2011. Complications following autologous bone graft harvesting from the iliac crest and using the RIA: a systematic review. *Injury*, 42 Suppl 2, S3-15.
- DING, W. 2016. Opportunities and challenges for the biodegradable magnesium alloys as next-generation biomaterials. *Regen Biomater*, 3, 79-86.
- ELGALI, I., OMAR, O., DAHLIN, C. & THOMSEN, P. 2017. Guided bone regeneration: materials and biological mechanisms revisited. *European journal of oral sciences*.
- FARKAS, R. & BECKS, T. 2005. Situation der Medizintechnik in Deutschland im internationalen Vergleich. *Aachen, Frankfurt*.
- FRODEL, J. L. 1999. The use of bone grafts in the upper craniomaxillofacial skeleton. *Facial* plastic surgery, 15, 25-32.
- GOMES, P. S., ZOMORODIAN, A., KWIATKOWSKI, L., LUTZE, R., BALKOWIEC, A., COLAÇO, B., PINHEIRO, V., FERNANDES, J. C., MONTEMOR, M. F. & FERNANDES, M. H. J. B. M. 2016. In vivo assessment of a new multifunctional coating architecture for improved Mg alloy biocompatibility. 11, 045007.
- HORNBERGER, H., VIRTANEN, S. & BOCCACCINI, A. R. 2012. Biomedical coatings on magnesium alloys a review. *Acta Biomater*, 8, 2442-55.
- JACOBI-GRESSER, E., HUESKER, K. & SCHUTT, S. 2013. Genetic and immunological markers predict titanium implant failure: a retrospective study. *Int J Oral Maxillofac Surg*, 42, 537-43.
- JIANG, H., WANG, J., CHEN, M. & LIU, D. 2017a. Biological activity evaluation of magnesium fluoride coated Mg-Zn-Zr alloy in vivo. *Mater Sci Eng C Mater Biol Appl*, 75, 1068-1074.
- JIANG, Y., WANG, B., JIA, Z., LU, X., FANG, L., WANG, K. & REN, F. J. J. O. B. M. R. P. A. 2017b. Polydopamine mediated assembly of hydroxyapatite nanoparticles and bone morphogenetic protein-2 on magnesium alloys for enhanced corrosion resistance and bone regeneration. 105, 2750-2761.
- JUNG, O., SMEETS, R., HARTJEN, P., SCHNETTLER, R., FEYERABEND, F., KLEIN, M., WEGNER, N., WALTHER, F., STANGIER, D., HENNINGSEN, A., RENDENBACH, C., HEILAND, M., BARBECK, M. & KOPP, A. 2019. Improved In Vitro Test Procedure for Full Assessment of the Cytocompatibility of Degradable Magnesium Based on ISO 10993-5/-12. Int J Mol Sci, 20.
- JUNG, O., SMEETS, R., KOPP, A., PORCHETTA, D., HIESTER, P., HEILAND, M., FRIEDRICH, R. E., PRECHT, C., HANKEN, H., GROBE, A. & HARTJEN, P. 2016. PEO-generated Surfaces Support Attachment and Growth of Cells In Vitro with No Additional Benefit for Micro-roughness in Sa (0.2-4 mum). *In Vivo*, 30, 27-33.
- JUNG, O., SMEETS, R., PORCHETTA, D., KOPP, A., PTOCK, C., MULLER, U., HEILAND, M., SCHWADE, M., BEHR, B., KROGER, N., KLUWE, L., HANKEN, H. & HARTJEN, P. 2015. Optimized in vitro procedure for assessing the cytocompatibility of magnesium-based biomaterials. *Acta Biomater*, 23, 354-63.
- KANG, M. H., JANG, T. S., KIM, S. W., PARK, H. S., SONG, J., KIM, H. E., JUNG, K. H. & JUNG, H. D. 2016. MgF2-coated porous magnesium/alumina scaffolds with improved strength, corrosion resistance, and biological performance for biomedical applications. *Mater Sci Eng C Mater Biol Appl*, 62, 634-42.

- KORZINSKAS, T., JUNG, O., SMEETS, R., STOJANOVIC, S., NAJMAN, S., GLENSKE, K., HAHN, M., WENISCH, S., SCHNETTLER, R. & BARBECK, M. 2018a. In Vivo Analysis of the Biocompatibility and Macrophage Response of a Non-Resorbable PTFE Membrane for Guided Bone Regeneration. *Int J Mol Sci*, 19, 2952.
- KORZINSKAS, T., JUNG, O., SMEETŠ, R., STOJANOVIC, S., NAJMAN, S., GLENSKE, K., HAHN, M., WENISCH, S., SCHNETTLER, R. & BARBECK, M. 2018b. In Vivo Analysis of the Biocompatibility and Macrophage Response of a Non-Resorbable PTFE Membrane for Guided Bone Regeneration. *Int J Mol Sci*, 19.
- LI, G., ZHANG, L., WANG, L., YUAN, G., DAI, K., PEI, J. & HAO, Y. 2018a. Dual modulation of bone formation and resorption with zoledronic acid-loaded biodegradable magnesium alloy implants improves osteoporotic fracture healing: An in vitro and in vivo study. *Acta Biomater*, 65, 486-500.
- LI, Z., SHIZHAO, S., CHEN, M., FAHLMAN, B. D., LIU, D., BI, H. J. M. S. & C, E. 2017. In vitro and in vivo corrosion, mechanical properties and biocompatibility evaluation of MgF2coated Mg-Zn-Zr alloy as cancellous screws. 75, 1268-1280.
- LI, Z., YANG, W., YU, Q., WU, Y., WANG, D., LIANG, J. & ZHOU, F. J. L. 2018b. New Method for the Corrosion Resistance of AZ31 Mg Alloy with a Porous Micro-Arc Oxidation Membrane as an Ionic Corrosion Inhibitor Container.
- LIU, W., YAN, Z., MA, X., GENG, T., WU, H. & LI, Z. J. M. 2018. Mg-MOF-74/MgF2 Composite Coating for Improving the Properties of Magnesium Alloy Implants: Hydrophilicity and Corrosion Resistance. 11, 396.
- MAKKAR, P., KANG, H. J., PADALHIN, A. R., PARK, I., MOON, B. G. & LEE, B. T. 2018. Development and properties of duplex MgF2/PCL coatings on biodegradable magnesium alloy for biomedical applications. *PLoS One*, 13, e0193927.
- MAO, L., YUAN, G., NIU, J., ZONG, Y. & DING, W. 2013. In vitro degradation behavior and biocompatibility of Mg-Nd-Zn-Zr alloy by hydrofluoric acid treatment. *Mater Sci Eng C Mater Biol Appl*, 33, 242-50.
- MILINKOVIC, I. & CORDARO, L. 2014. Are there specific indications for the different alveolar bone augmentation procedures for implant placement? A systematic review. *International journal of oral and maxillofacial surgery*, 43, 606-625.
- NEACSU, P., STARAS, A. I., VOICU, S. I., IONASCU, I., SOARE, T., UZUN, S., COJOCARU, V. D., PANDELE, A. M., CROITORU, S. M., MICULESCU, F., COTRUT, C. M., DAN, I. & CIMPEAN, A. 2017. Characterization and In Vitro and In Vivo Assessment of a Novel Cellulose Acetate-Coated Mg-Based Alloy for Orthopedic Applications. *Materials (Basel)*, 10, 686.
- PERUMAL, G., RAMASAMY, B., DOBLE, M. J. C. & BIOINTERFACES, S. B. 2018. Nanostructure coated AZ31 magnesium cylindrical mesh cage for potential long bone segmental defect repair-in vitro degradation and cytocompatibility studies.
- RAWASHDEH, M. A. & TELFAH, H. 2008. Secondary alveolar bone grafting: the dilemma of donor site selection and morbidity. *Br J Oral Maxillofac Surg*, 46, 665-70.
- SHEIKH, Z., HAMDAN, N., IKEDA, Y., GRYNPAS, M., GANSS, B. & GLOGAUER, M. 2017. Natural graft tissues and synthetic biomaterials for periodontal and alveolar bone reconstructive applications: a review. *Biomater Res*, 21, 9.
- SINDET-PEDERSEN, S. & ENEMARK, H. 1990. Reconstruction of alveolar clefts with mandibular or iliac crest bone grafts: a comparative study. *J Oral Maxillofac Surg*, 48, 554-8; discussion 559-60.
- SMEETS, R., JUNG, O., HANKEN, H., HARTJEN, P., AL DAM, A., GRÖBE, A., HEILAND, M., GOSAU, M., ROTHAMEL, D. & SCHLEE, M. 2014. Was können regenerative Materialien in der Zahnmedizin leisten–und wo sind die Grenzen. *DZZ*, 69, 708-721.
- STAIGER, M. P., PIETAK, A. M., HUADMAI, J. & DIAS, G. 2006. Magnesium and its alloys as orthopedic biomaterials: a review. *Biomaterials*, 27, 1728-34.
- TOLSTUNOV, L., HAMRICK, J. F. E., BROUMAND, V., SHILO, D. & RACHMIEL, A. 2019. Bone Augmentation Techniques for Horizontal and Vertical Alveolar Ridge Deficiency in Oral Implantology. *Oral Maxillofac Surg Clin North Am*, 31, 163-191.

- WITTE, F., FISCHER, J., NELLESEN, J., CROSTACK, H. A., KAESE, V., PISCH, A., BECKMANN, F. & WINDHAGEN, H. 2006. In vitro and in vivo corrosion measurements of magnesium alloys. *Biomaterials*, 27, 1013-8.
- WITTE, F., KAESE, V., HAFERKAMP, H., SWITZER, E., MEYER-LINDENBERG, A., WIRTH, C. J. & WINDHAGEN, H. 2005. In vivo corrosion of four magnesium alloys and the associated bone response. *Biomaterials*, 26, 3557-63.
- YU, W., ZHAO, H., DING, Z., ZHANG, Z., SUN, B., SHEN, J., CHEN, S., ZHANG, B., YANG, K., LIU, M., CHEN, D. & HE, Y. 2017. In vitro and in vivo evaluation of MgF2 coated AZ31 magnesium alloy porous scaffolds for bone regeneration. *Colloids Surf B Biointerfaces*, 149, 330-340.
- ZHANG, J., KONG, N., NIU, J., SHI, Y., LI, H., ZHOU, Y. & YUAN, G. J. J. O. M. S. M. I. M. 2014. Influence of fluoride treatment on surface properties, biodegradation and cytocompatibility of Mg–Nd–Zn–Zr alloy. 25, 791-799.
- ZHAO, D., WITTE, F., LU, F., WANG, J., LI, J. & QIN, L. 2017. Current status on clinical applications of magnesium-based orthopaedic implants: A review from clinical translational perspective. *Biomaterials*, 112, 287-302.
- ZHAO, N., WORKMAN, B. & ZHU, D. 2014. Endothelialization of novel magnesium-rare earth alloys with fluoride and collagen coating. *Int J Mol Sci*, 15, 5263-76.
- ZHENG, Y., GU, X. & WITTE, F. 2014. Biodegradable metals. *Materials Science and Engineering: R: Reports,* 77, 1-34.
- ZOMORODIAN, A., GARCIA, M. P., MOURA E SILVA, T., FERNANDES, J. C., FERNANDES, M. H. & MONTEMOR, M. F. 2013. Corrosion resistance of a composite polymeric coating applied on biodegradable AZ31 magnesium alloy. *Acta Biomater*, 9, 8660-70.

## Erklärung des Eigenanteils an der Publikation

Mein Anteil an der vorliegenden Publikationsdissertation umfasst die Mitwirkung in der Durchführung der *in vivo*-Experimente, die Explantation und Assistenz bei der histologischen Aufarbeitung der Präparate, die histomorphometrische Auswertung, die Auswertung der digitalen Volumentomographie und der Kontaktradiographie, die Literaturrecherche, sowie die Anfertigung der dokumentarischen Fotografien (intraoperativ, histopathologische Auswertung, Analysemethodik). Für die Auswertung der digitalen Volumentomographie, sowie der Kontaktradiographie habe ich selbstständig ein digitales Analysemodul entwickelt und angewendet. Die histopathologische Analyse und das Verfassen des Manuskripts erfolgte in Abstimmung mit meinen Betreuern. Die Durchführung und Auswertung der *in vitro*-Experimente erfolgte durch Dr. med. Ole Jung.

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Lebenslauf wurde aus datenschutzrechtlichen Gründen entfernt.

## **Eidesstattliche Versicherung**

Ich versichere ausdrücklich, dass ich die Arbeit selbständig und ohne fremde Hilfe verfasst, andere als die von mir angegebenen Quellen und Hilfsmittel nicht benutzt und die aus den benutzten Werken wörtlich oder inhaltlich entnommenen Stellen einzeln nach Ausgabe (Auflage und Jahr des Erscheinens), Band und Seite des benutzten Werkes kenntlich gemacht habe.

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Ich erkläre mich einverstanden, dass meine Dissertation vom Dekanat der Medizinischen Fakultät mit einer gängigen Software zur Erkennung von Plagiaten überprüft werden kann.

Unterschrift: .....